



• Diffuse Scattering



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- Modulated structures



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- Lattice vibrations



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- Powder diffraction



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Reading Assignment: Chapter 6.1-6.2



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Homework Assignment #04: Chapter 4: 2,4,6,7,10 due Friday, October 18, 2024



- Diffuse Scattering
- Modulated structures
- Lattice vibrations
- Powder diffraction
- Pair distribution function
- Bragg & Laue geometries

Reading Assignment: Chapter 6.1-6.2

Homework Assignment #04: Chapter 4: 2,4,6,7,10 due Friday, October 18, 2024 Homework Assignment #05: Chapter 5: 1,3,7,9,10 due Monday, October 28, 2024



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This effect gets larger for larger momentum transfers





"Critical role of a buried interface in the Stranski-Krastanov growth of metallic nanocrystals: Quantum size effects in Ag/Si(111)-(7×7)," Y. Chen et al. Phys. Rev. Lett. 114, 035501 (2015).

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Understanding the process of surface wetting during thin film deposition is crucial to the semiconductor industry

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Ag was thermally evaporated on the surface and both reflectivity measurements of the surface and CTR measurements of the Ag (001) growth layer were performed



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Reflectivity "sees" the entire surface while CTR measures only the incommensurate Ag crystalline layer on the surface

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Modeling shows that the islands are displaced from the surface

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The exceptional stability of the three layer islands is consistent with quantum confinement effects that drive the growth process

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The islands thus have a weak interaction with the substrate compared to the wetting layer

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Modulated structures



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However, it is common to see structures where the positions of the atoms is modulated (e.g. charge density waves, magnetic lattices, etc.) according to $x_n = an + u \cos(qan)$, where: *a* is the lattice parameter, *u* is the amplitude of the displacement, and $q = 2\pi/\lambda_m$ is the wave vector of the modulation.

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If λ_m is a multiple or a rational fraction of a, it is called a commensurate modulation but if $\lambda_m = ca$, where c is an irrational number, then it is an incommensurate modulation.

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the diffraction pattern has main Bragg peaks plus satellite peaks

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If the modulation of the structure is a multiple of the lattice parameter, the modulation is simply a superlattice and the actual lattice parameter will be changed.





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In 2011 Shechtman was awarded the Nobel Prize in Chemistry







"Metallic phase with long-range orientational order and no translational symmetry," D. Shechtman, I. Blech, D. Gratias, and J.W. Cahn, Phys. Rev. Lett. 53, 1951-1953 (1984)

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V

The electron micrographs show that there must be long range order to be able to get such sharp diffraction peaks

The 5-fold symmetry is evident in the 10 spots surrounding the center of the left image and the pentagonal arrangements of atoms in the image on the right.



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Other groups have discovered stable icosahedral phases with three and two elements.

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Quasicrystal diffraction patterns

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The $AI_{65}Cu_{20}Fe_{15}$ system was one of the first stable quasicrystals to be discovered. Later discovery of stable quasicrystals in the Ta-Te, Cd-Ca, and Cd-Yb systems enabled large crystals to be grown.

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The last term is a time average which can be simplified by first taking the scalar product, $\vec{Q} \cdot \vec{u}_n = u_{Qn}$ to project the displacement along the scattering vector, then applying the Baker-Hausdorff theorem, $\langle e^{ix} \rangle = e^{-\langle x^2 \rangle/2}$

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$$I = \left\langle \sum_{m} f(\vec{Q}) e^{i\vec{Q}\cdot(\vec{R}_{m}+\vec{u}_{m})} \sum_{n} f^{*}(\vec{Q}) e^{-i\vec{Q}\cdot(\vec{R}_{n}+\vec{u}_{n})} \right\rangle$$
$$= \sum_{m} \sum_{n} f(\vec{Q}) f^{*}(\vec{Q}) e^{i\vec{Q}\cdot(\vec{R}_{m}-\vec{R}_{n})} \left\langle e^{i\vec{Q}\cdot(\vec{u}_{m}-\vec{u}_{n})} \right\rangle$$

The last term is a time average which can be simplified by first taking the scalar product, $\vec{Q} \cdot \vec{u}_n = u_{Qn}$ to project the displacement along the scattering vector, then applying the Baker-Hausdorff theorem, $\langle e^{ix} \rangle = e^{-\langle x^2 \rangle/2}$

$$\left\langle e^{i\vec{Q}\cdot(\vec{u}_m-\vec{u}_n)} \right\rangle = \left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle$$

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$$\left\langle e^{i\vec{Q}\cdot(\vec{u}_m-\vec{u}_n)} \right\rangle = \left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle = e^{-\langle Q^2(u_{Qm}-u_{Qn})^2 \rangle/2}$$

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$$\left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle = e^{-Q^2 \langle u_{Qm}^2 \rangle/2} e^{-Q^2 \langle u_{Qn}^2 \rangle/2} e^{Q^2 \langle u_{Qm}u_{Qn} \rangle}$$



$$ig\langle e^{iQ(u_{Qm}-u_{Qn})}ig
angle = e^{-Q^2\langle u_{Qm}^2
angle/2}e^{-Q^2\langle u_{Qn}^2
angle/2}e^{Q^2\langle u_{Qm}u_{Qn}
angle}
onumber \ = e^{-Q^2\langle u_Q^2
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angle = e^{-Q^2\langle u_{Qm}^2
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angle}e^{Q^2\langle u_{Qm}u_{Qn}
angle} = e^{-2M}e^{Q^2\langle u_{Qm}u_{Qn}
angle}$$



$$\left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle = e^{-Q^2 \langle u_{Qm}^2 \rangle / 2} e^{-Q^2 \langle u_{Qn}^2 \rangle / 2} e^{Q^2 \langle u_{Qm} u_{Qn} \rangle}$$
$$= e^{-Q^2 \langle u_{Q}^2 \rangle} e^{Q^2 \langle u_{Qm} u_{Qn} \rangle} = e^{-2M} e^{Q^2 \langle u_{Qm} u_{Qn} \rangle} = e^{-2M} \left[1 + e^{Q^2 \langle u_{Qm} u_{Qn} \rangle} - 1 \right]$$



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$$I = \sum_{m} \sum_{n} f(\vec{Q}) e^{-M} e^{i\vec{Q}\cdot\vec{R}_{m}} f^{*}(\vec{Q}) e^{-M} e^{-i\vec{Q}\cdot\vec{R}_{n}} + \sum_{m} \sum_{n} f(\vec{Q}) e^{-M} e^{i\vec{Q}\cdot\vec{R}_{m}} f^{*}(\vec{Q}) e^{-M} e^{-i\vec{Q}\cdot\vec{R}_{n}} \left[e^{Q^{2} \langle u_{Qm} u_{Qn} \rangle} - 1 \right]$$



$$\left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle = e^{-Q^2 \langle u_{Qm}^2 \rangle / 2} e^{-Q^2 \langle u_{Qn}^2 \rangle / 2} e^{Q^2 \langle u_{Qm}u_{Qn} \rangle}$$
$$= e^{-Q^2 \langle u_Q^2 \rangle} e^{Q^2 \langle u_{Qm}u_{Qn} \rangle} = e^{-2M} e^{Q^2 \langle u_{Qm}u_{Qn} \rangle} = e^{-2M} \left[1 + e^{Q^2 \langle u_{Qm}u_{Qn} \rangle} - 1 \right]$$

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$$+ \sum_{m} \sum_{n} f(\vec{Q}) e^{-M} e^{i\vec{Q}\cdot\vec{R}_{m}} f^{*}(\vec{Q}) e^{-M} e^{-i\vec{Q}\cdot\vec{R}_{n}} \left[e^{Q^{2}\langle u_{Qm}u_{Qn}\rangle} - 1 \right]$$

The first term is just the elastic scattering from the lattice with the addition of the term $e^{-M} = e^{-Q^2 \langle u_Q^2 \rangle/2}$, called the Debye-Waller factor.

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$$\left\langle e^{iQ(u_{Qm}-u_{Qn})} \right\rangle = e^{-Q^2 \langle u_{Qm}^2 \rangle / 2} e^{-Q^2 \langle u_{Qn}^2 \rangle / 2} e^{Q^2 \langle u_{Qm}u_{Qn} \rangle}$$
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The first term is just the elastic scattering from the lattice with the addition of the term $e^{-M} = e^{-Q^2 \langle u_Q^2 \rangle/2}$, called the Debye-Waller factor.

The second term is the Thermal Diffuse Scattering and actually increases with mean squared displacement.

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$$I^{TDS} = \sum_{m} \sum_{n} f(\vec{Q}) e^{-M} e^{i\vec{Q}\cdot\vec{R}_{m}} f^{*}(\vec{Q}) e^{-M} e^{-i\vec{Q}\cdot\vec{R}_{n}} \left[e^{Q^{2} \langle u_{Qm}u_{Qn} \rangle} - 1 \right]$$



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The TDS has a width determined by the correlated displacement of atoms which is much broader than a Bragg peak.



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These correlated motions are just phonons.



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"Determination of phonon dispersions from x-ray transmission scattering: The example of silicon," M. Holt, et al. *Phys. Rev. Lett.* **83**, 3317 (1999).



incident beam along (100)



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incident beam along (111)



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dotted line from this measurement

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$$\mathcal{F}^{u.c.} = \sum_j f_j(\vec{Q}) e^{-M_j} e^{i \vec{Q} \cdot \vec{r}_j}$$



$$egin{aligned} \mathcal{F}^{u.c.} &= \sum_{j} f_{j}(ec{Q}) e^{-M_{j}} e^{iec{Q}.ec{r}_{j}} \ M_{j} &= rac{1}{2} Q^{2} \langle u_{Qj}^{2}
angle \end{aligned}$$



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angle = rac{1}{2} \left(rac{4\pi}{\lambda}
ight)^{2} \sin^{2} heta \langle u_{Qj}^{2}
angle \end{aligned}$$



$$B_T^j = 8\pi^2 \langle u_{Qj}^2 \rangle$$

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For crystals with several different types of atoms, we generalize the unit cell scattering factor.

$$B_T^j = 8\pi^2 \langle u_{Qj}^2 \rangle$$

for isotropic atomic vibrations

1

$$\langle u^2 \rangle = \langle u_x^2 + u_y^2 + u_z^2 \rangle$$

= $3 \langle u_x^2 \rangle = 3 \langle u_Q^2 \rangle$



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$$\begin{split} \langle u^2 \rangle &= \langle u_x^2 + u_y^2 + u_z^2 \rangle \\ &= 3 \langle u_x^2 \rangle = 3 \langle u_Q^2 \rangle \end{split}$$

$$F^{u.c.} = \sum_{j} f_{j}(\vec{Q}) e^{-M_{j}} e^{i\vec{Q}\cdot\vec{r}_{j}}$$
$$M_{j} = \frac{1}{2} Q^{2} \langle u_{Qj}^{2} \rangle = \frac{1}{2} \left(\frac{4\pi}{\lambda}\right)^{2} \sin^{2}\theta \langle u_{Qj}^{2} \rangle$$
$$M_{j} = B_{T}^{j} \left(\frac{\sin\theta}{\lambda}\right)^{2}$$
$$B_{T}^{iso} = \frac{8\pi^{2}}{3} \langle u^{2} \rangle$$



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In general, Debye-Waller factors can be anisotropic

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The Debye model can be used to compute B_T by integrating a linear phonon dispersion relation up to a cutoff frequency, ω_D , called the Debye frequency.

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The Debye model can be used to compute B_T by integrating a linear phonon dispersion relation up to a cutoff frequency, ω_D , called the Debye frequency.

$$B_T = rac{6h^2}{m_A k_B \Theta} \left[rac{\phi(\Theta/T)}{\Theta/T} + rac{1}{4}
ight]$$





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$$B_{T} = \frac{6h^{2}}{m_{A}k_{B}\Theta} \left[\frac{\phi(\Theta/T)}{\Theta/T} + \frac{1}{4}\right]$$
$$\phi(x) = \frac{1}{x} \int_{0}^{\Theta/T} \frac{\xi}{e^{\xi} - 1} d\xi$$





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$$B_{\mathcal{T}}[\text{\AA}^2] = \frac{11492\text{T}[\text{K}]}{\text{A}\Theta^2[\text{K}^2]}\phi(\Theta/\text{T}) + \frac{2873}{\text{A}\Theta[\text{K}]}$$





| | A | Θ | $B_{4.2}$ | B ₇₇ | B_{293} |
|----------|------|------|-----------|-----------------|-----------|
| | | (K) | | (Ų) | |
| C* | 12 | 2230 | 0.11 | 0.11 | 0.12 |
| AI | 27 | 428 | 0.25 | 0.30 | 0.72 |
| Cu | 63.5 | 343 | 0.13 | 0.17 | 0.47 |
| *diamond | | | | | |

 $B_T = \frac{11492T}{A\Theta^2}\phi(\Theta/T) + \frac{2873}{A\Theta}$

diamond



| | Α | Θ | B _{4.2} | B ₇₇ | B ₂₉₃ | |
|----------|------|------|------------------|-----------------|------------------|--|
| | | (K) | | $(Å^2)$ | | |
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$$B_{\mathcal{T}} = rac{11492\,T}{\mathcal{A}\Theta^2} \phi(\Theta/\,T) + rac{2873}{\mathcal{A}\Theta}$$

diamond is very stiff and B_T does not vary much with temperature



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 $B_{T} = \frac{11492T}{A\Theta^{2}}\phi(\Theta/T) + \frac{2873}{A\Theta}$

diamond is very stiff and B_T does not vary much with temperature

copper has a much lower Debye temperature and a wider variation of thermal factor with temperature



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Temperature (K)

Beamline 11BM at the APS





"A dedicated powder diffraction beamline at the Advanced Photon Source: Commissioning and early operational results," J. Wang et al. Rev. Sci. Instrum. 79, 085105 (2008).

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Beamline 11BM at the APS





2D detectors have limited angular resolution, for high resolution routine powder diffraction, beamlines such as 11BM are ideal

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Beamline 11BM at the APS





2D detectors have limited angular resolution, for high resolution routine powder diffraction, beamlines such as 11BM are ideal

The initial collimating mirror makes the beam more parallel and then it is focused horizontally and vertically to the sample

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[&]quot;A dedicated powder diffraction beamline at the Advanced Photon Source: Commissioning and early operational results," J. Wang et al. Rev. Sci. Instrum. 79, 085105 (2008).



"A twelve-analyzer detector system for high resolution powder diffraction," P.L. Lee et al. J. Synch. Rad. 15, 427-432

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High throughput is obtained using a robot arm to change samples

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High throughput is obtained using a robot arm to change samples

The sample is mounted on a rotating spindle at the center of the goniometer

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High throughput is obtained using a robot arm to change samples

The sample is mounted on a rotating spindle at the center of the goniometer

High resolution is achieved with a 12 crystal analyzer system which is rotated on the main circle of the goniometer

"A twelve-analyzer detector system for high resolution powder diffraction," P.L. Lee et al. *J. Synch. Rad.* **15**, 427-432

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Each of the 12 analyzer crystals is tuned to the desired scattering energy and as the entire assembly is scanned, all twelve banks are collecting data and then are merged

"A twelve-analyzer detector system for high resolution powder diffraction," P.L. Lee et al. J. Synch. Rad. 15, 427-432

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The analyzer and robot arm



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The analyzer and robot arm



Samples are in Kapton capillaries and magnetic bases for remote mounting



"A twelve-analyzer detector system for high resolution powder diffraction," P.L. Lee et al. J. Synch. Rad. 15, 427-432

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Data from high resolution LaB₆ standard



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dif.

Data from high resolution LaB₆ standard



High resolution data with high count rates can be obtained out to very high angles with a wave-length of $\lambda \approx 0.5$ Å.

"A dedicated powder diffraction beamline at the Advanced Photon Source: Commissioning and early operational results," J. Wang et al. Rev. Sci. Instrum. 79, 085105 (2008).

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Refinement of SiO_2 and AI_2O_3





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