

Today's outline - October 26, 2021





- Dumond diagrams & monochromators

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- Photoelectric absorption

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- X-ray absorption spectroscopy

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Reading Assignment: Chapter 7.2-3

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Reading Assignment: Chapter 7.2-3

Homework Assignment #05:

Chapter 5: 1,3,7,9,10

due Tuesday, November 02, 2021

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- Dumond diagrams & monochromators
- Photoelectric absorption
- X-ray absorption spectroscopy

Reading Assignment: Chapter 7.2-3

Homework Assignment #05:

Chapter 5: 1,3,7,9,10

due Tuesday, November 02, 2021

Homework Assignment #06:

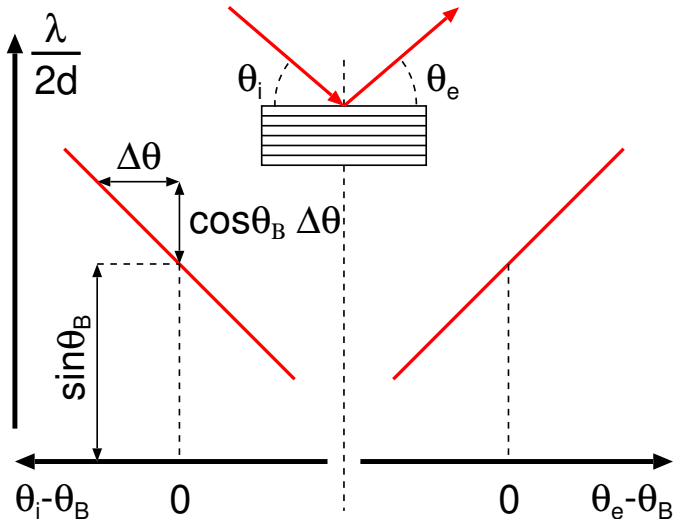
Chapter 6: 1,6,7,8,9

due Tuesday, November 16, 2021

Dumond diagram: no Darwin width



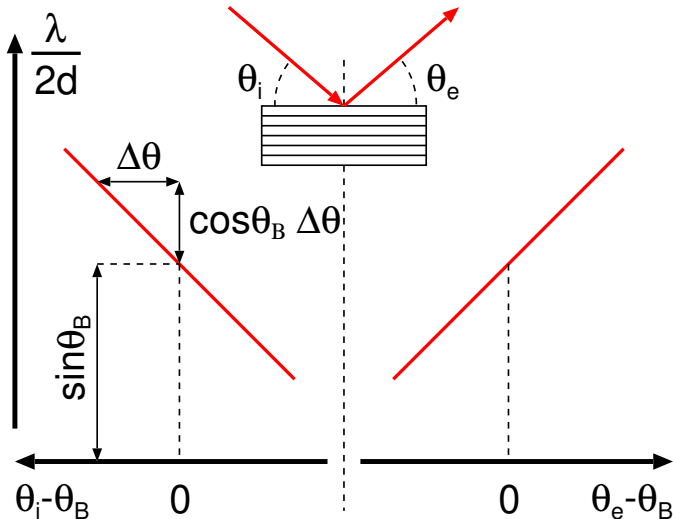
Transfer function of an optical element parametrized by angle and wavelength.



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Transfer function of an optical element parametrized by angle and wavelength. Here Darwin width is ignored.

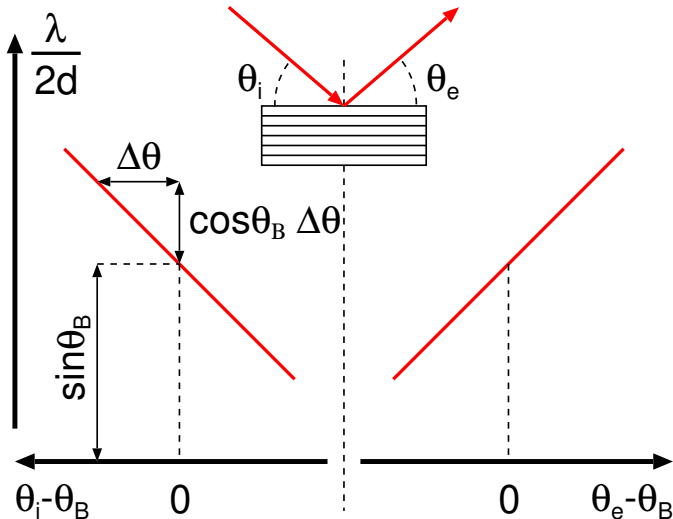


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for small angular deviations $\sin \theta$ is linear with a slope of $\cos \theta_B$



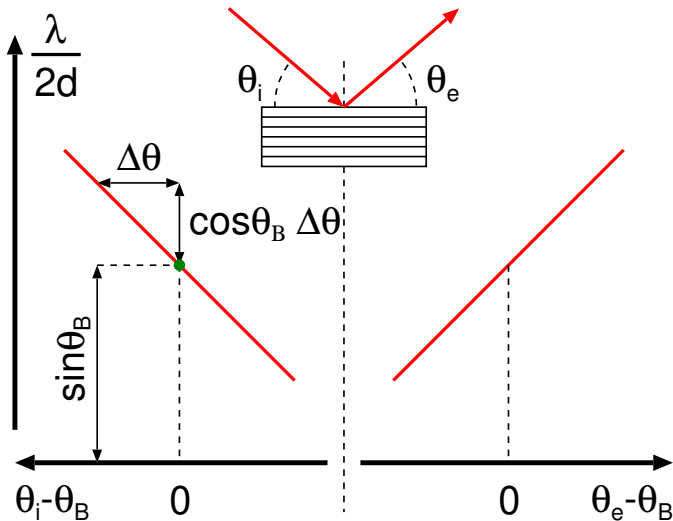
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non-zero diffracted beam only for points on the line



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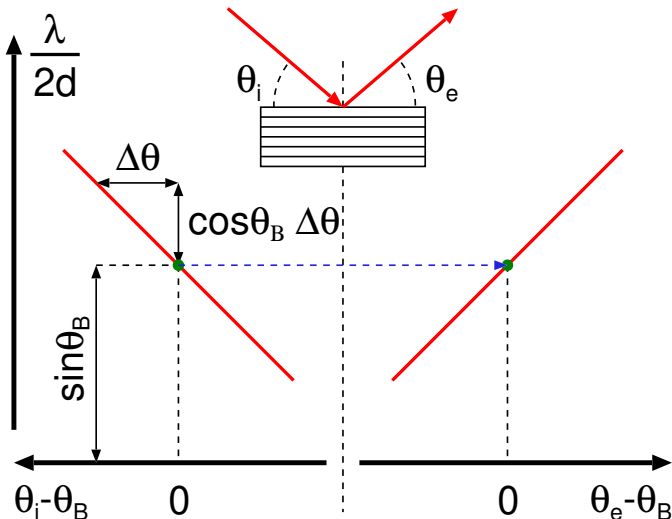


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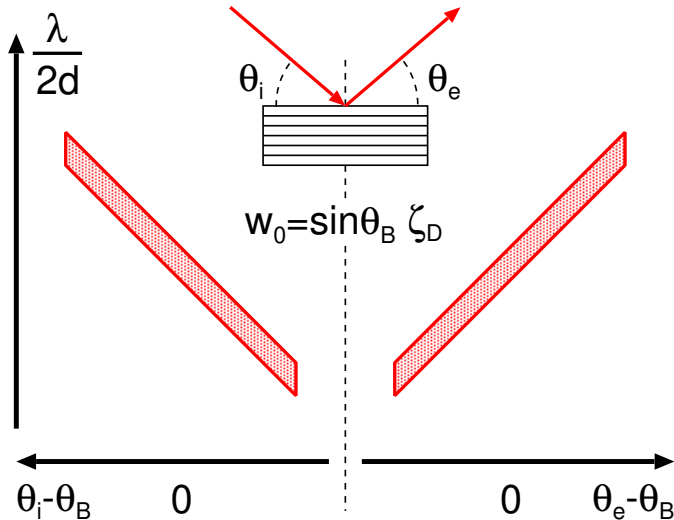
for small angular deviations $\sin \theta$ is linear with a slope of $\cos \theta_B$

non-zero diffracted beam only for points on the line

a horizontal line transfers input to output beam characteristics

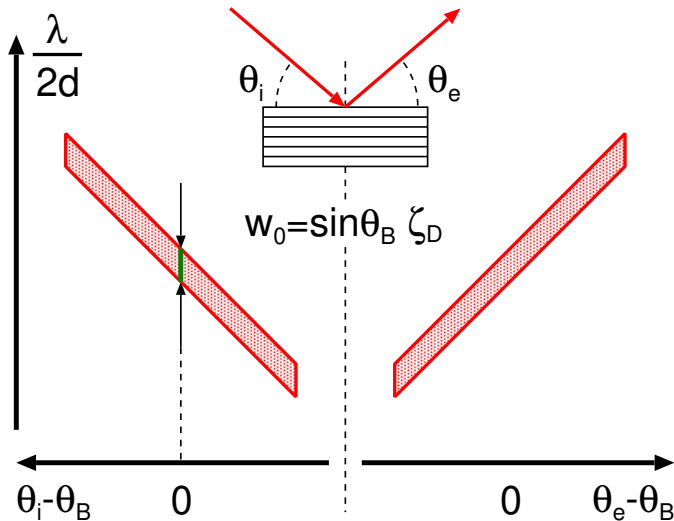


Dumond diagram: symmetric Bragg



If Darwin width is included, the Bragg condition is represented by a **box**

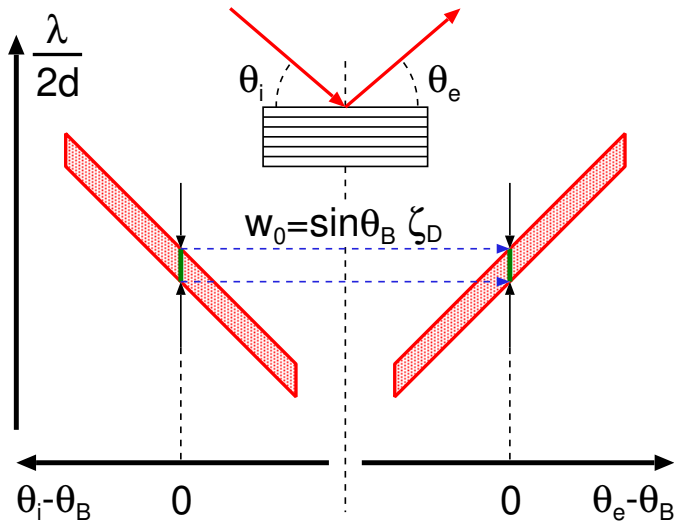
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for a perfectly collimated (no angular divergence) input beam, a **bandwidth** of radiation is accepted by the crystal

Dumond diagram: symmetric Bragg

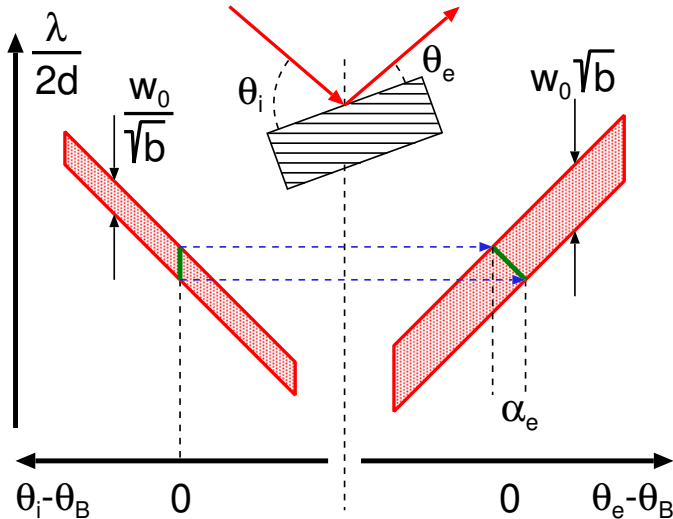


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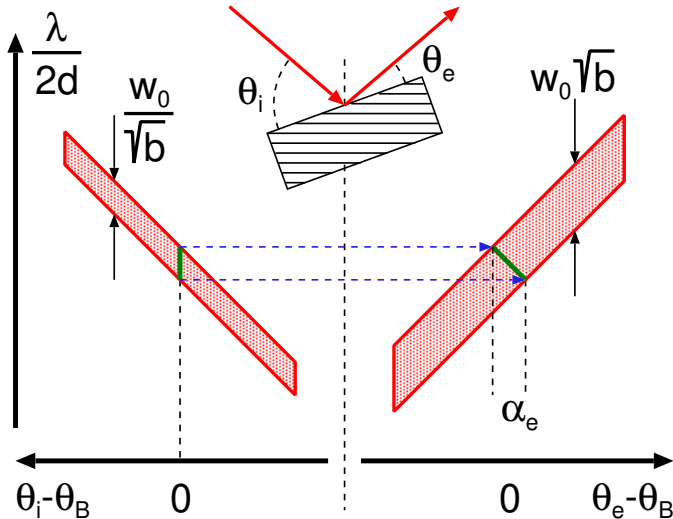
this input bandwidth is **transferred** to a similar output bandwidth which is also collimated

Dumond diagram: asymmetric Bragg



For an asymmetric crystal, the output beam is no longer collimated but acquires a divergence α_e

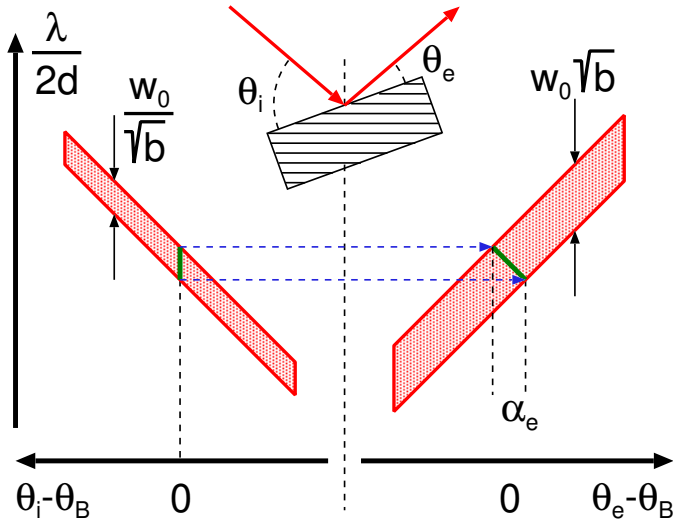
Dumond diagram: asymmetric Bragg



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a perfectly collimated input beam transfers to an output beam that has an angular divergence which depends on the asymmetry factor b

Dumond diagram: asymmetric Bragg

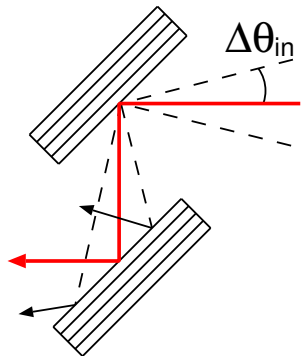


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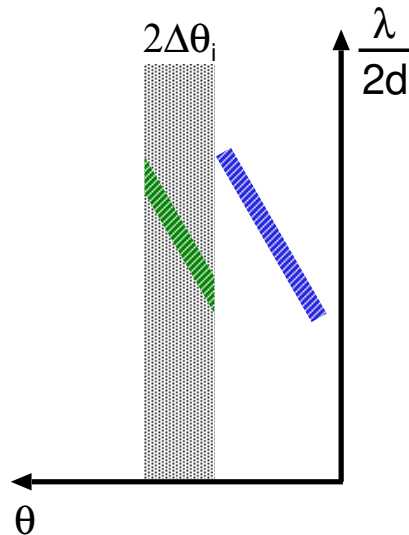
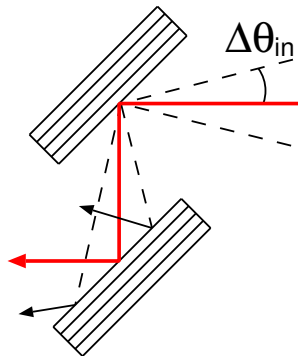
a perfectly collimated input beam transfers to an output beam that has an angular divergence which depends on the asymmetry factor b

this is in addition to a compression (in this case) of the beam height (Liouville's theorem!)

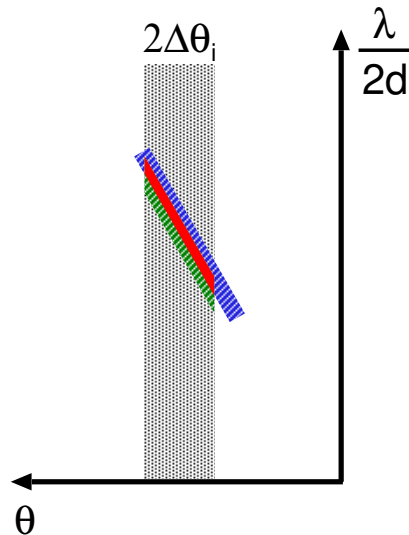
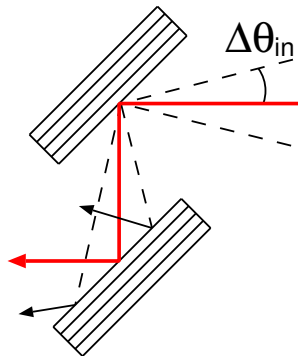
Double crystal monochromator: Non-dispersive



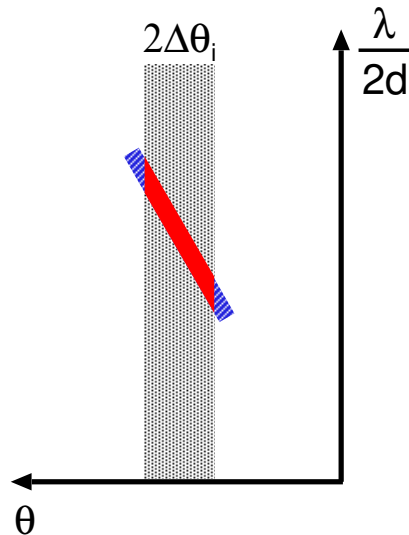
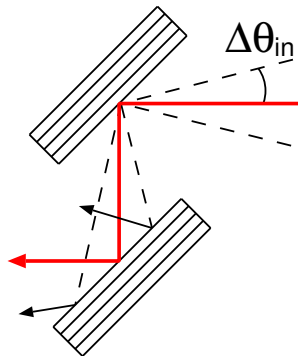
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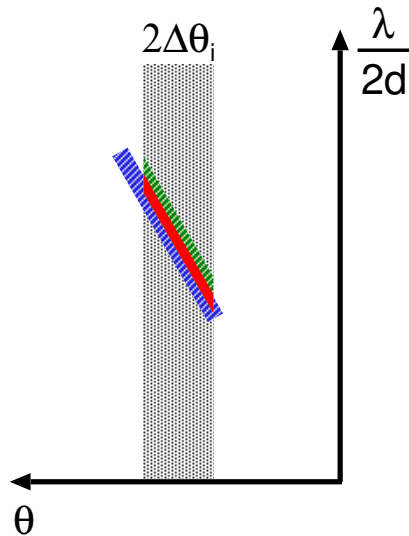
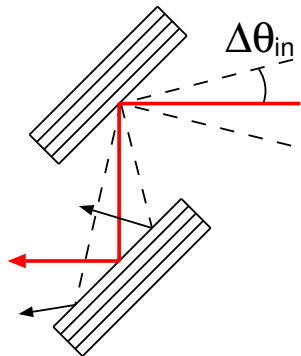
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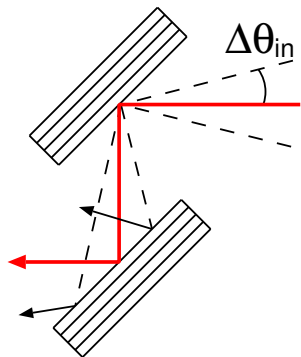
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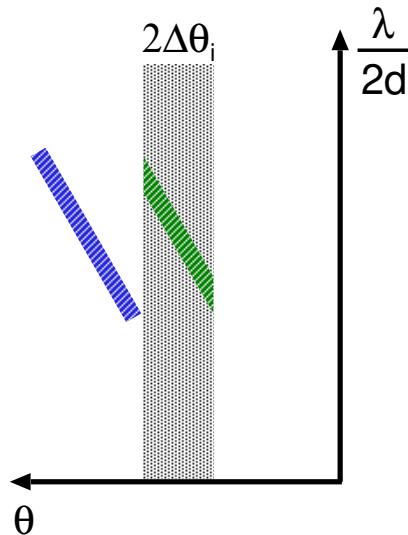
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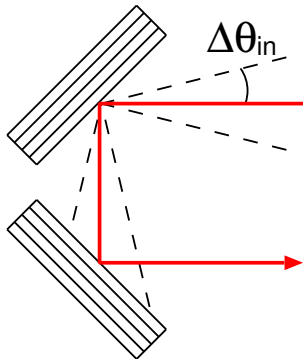
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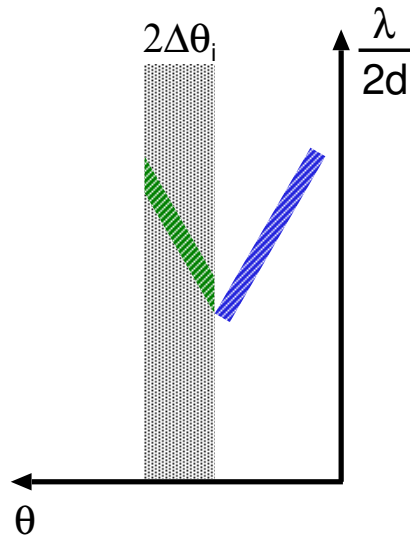
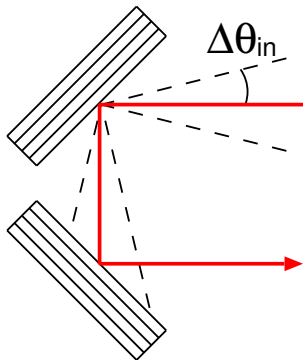
the transfer functions of the two crystals match and full bandwidth and divergence is preserved, giving a triangle intensity curve



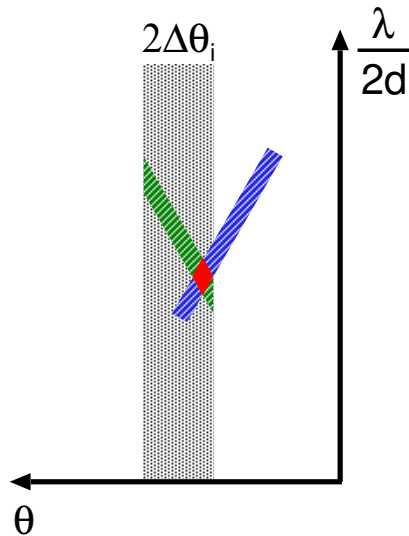
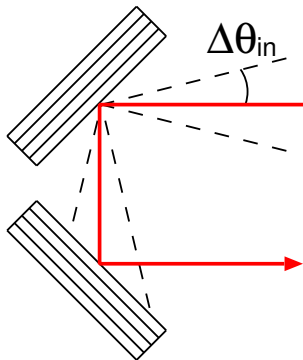
Double crystal monochromators: Dispersive



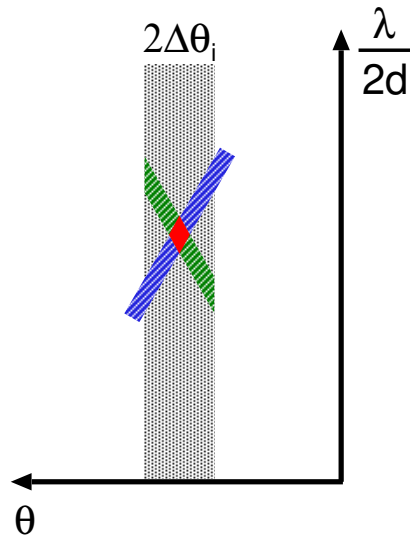
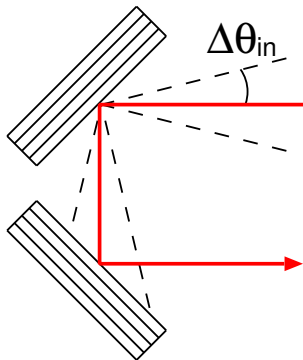
Double crystal monochromators: Dispersive



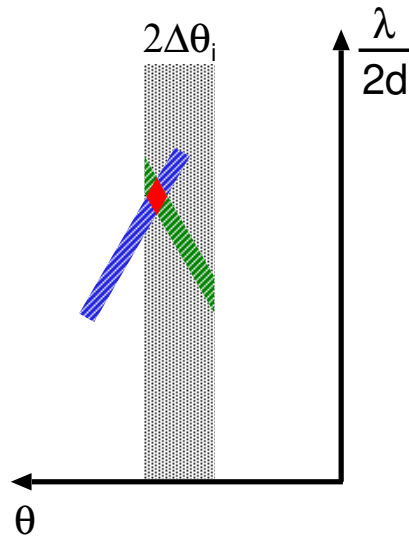
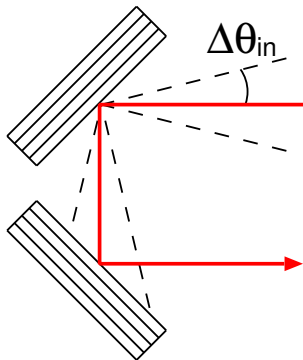
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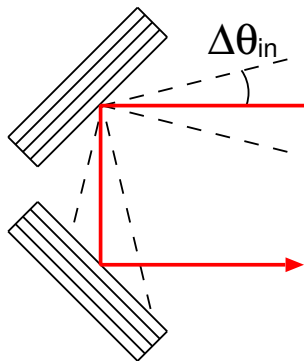
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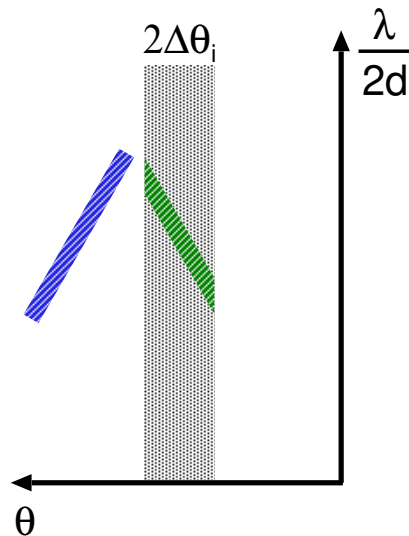
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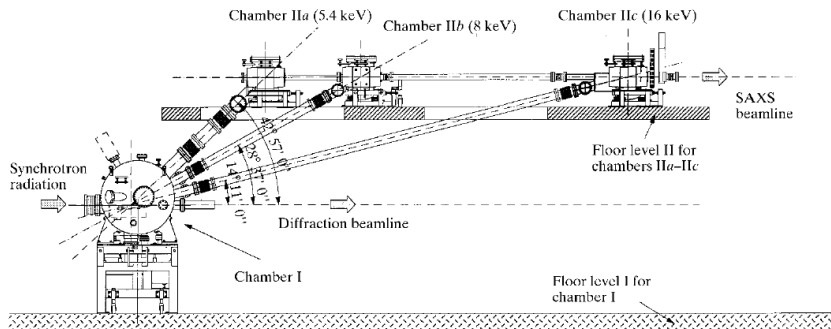
Double crystal monochromators: Dispersive



the transfer function matches only in a small energy band that varies with angle of the second crystal, mapping out the Darwin curve of the first crystal

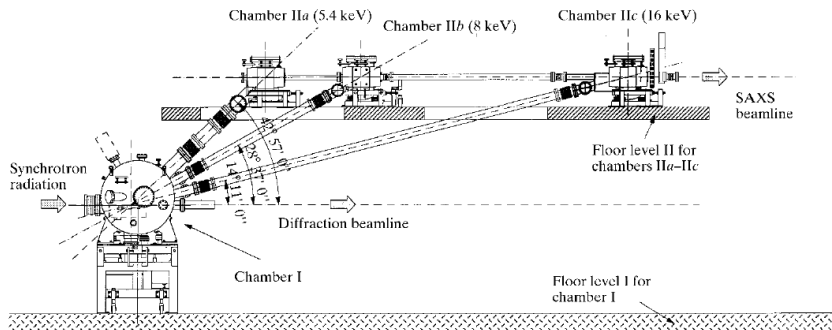


Asymmetric monochromator at ELETTRA



"High-throughput asymmetric double-crystal monochromator of the SAXS beamline at ELETTRA," S. Bernstorff, H. Amentisch, and P. Laggner, *J. Synchrotron Rad.* **5**, 1215-1221 (1998).

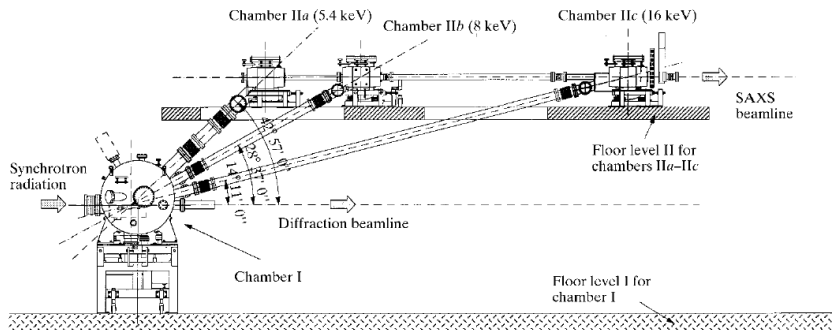
Asymmetric monochromator at ELETTRA



The SAXS beamline at ELETTRA has asymmetric cut crystals with 2° grazing incidence in order to spread the heat load

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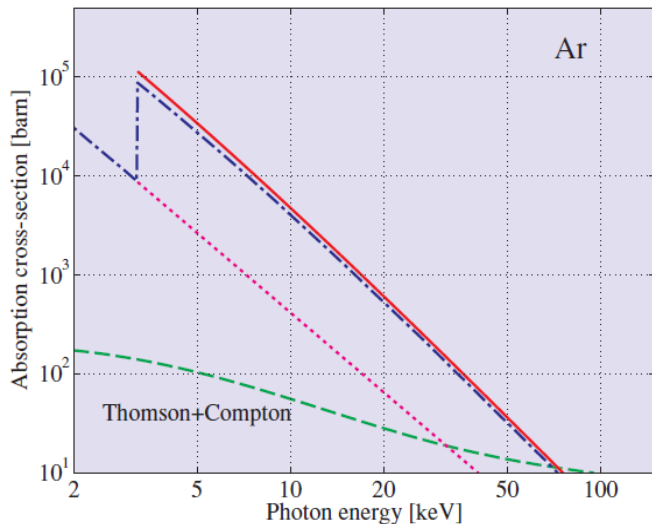


The SAXS beamline at ELETTRA has asymmetric cut crystals with 2° grazing incidence in order to spread the heat load

The three crystals are set for single energies of 5.6, 8.0, and 16 keV with a vertical displacement of 1.5 m and asymmetry parameter, b , of 0.053, 0.078, and 0.17, respectively

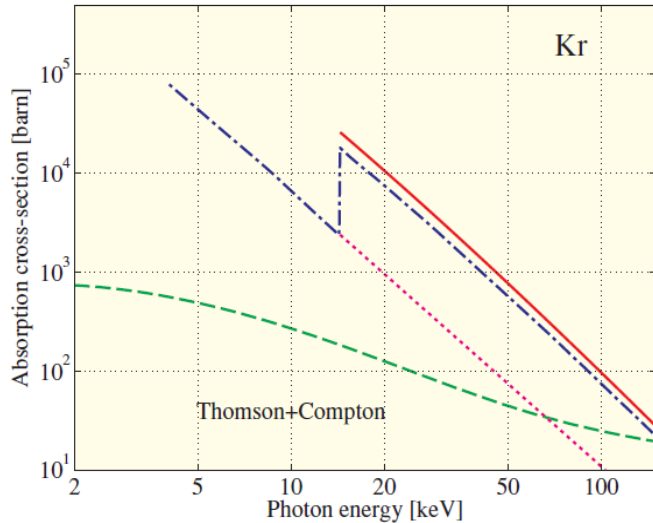
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Total cross section



The total cross-section for photon “absorption” includes elastic (or coherent) scattering, Compton (inelastic) scattering, and photoelectric absorption.

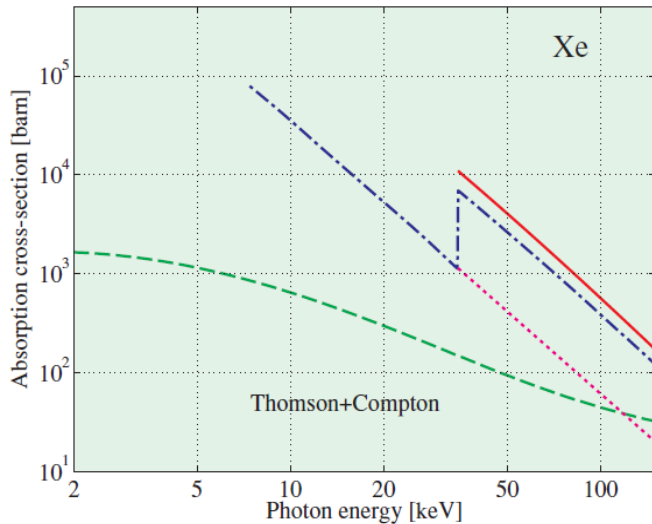
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Characteristic absorption jumps depend on the element

Total cross section



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Characteristic absorption jumps depend on the element

These quantities vary significantly over many decades but can easily be put on an equal footing.

Scaled absorption



$$T = \frac{I}{I_0} = e^{-\mu z}$$

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$$\mu = \frac{\rho_m N_A}{M} \sigma_a$$

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σ_a can be scaled for different elements
by E^3/Z^4 and plotted together

Scaled absorption



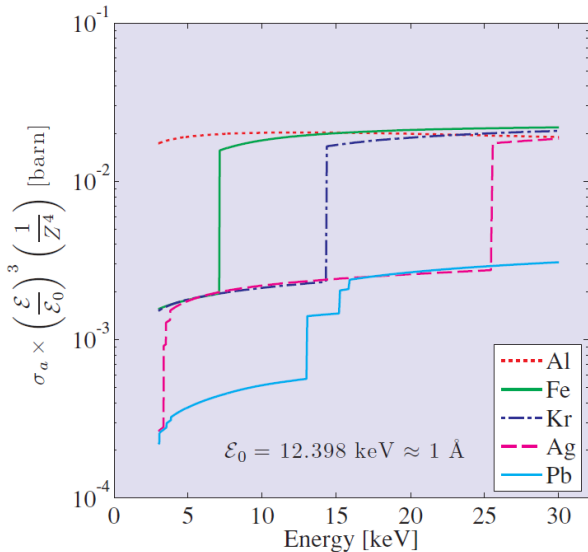
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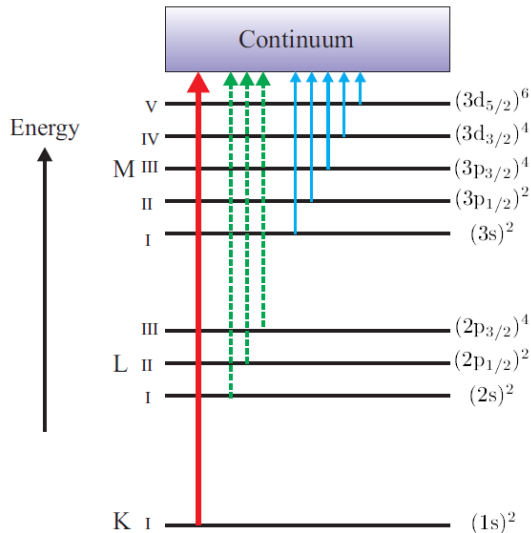
remarkably, all values lie on a common curve above the K edge and between the L and K edges and below the L edge



Absorption edge nomenclature



The states are labeled according to the principal, orbital angular momentum, and total angular momentum quantum numbers, n , l , and j , respectively

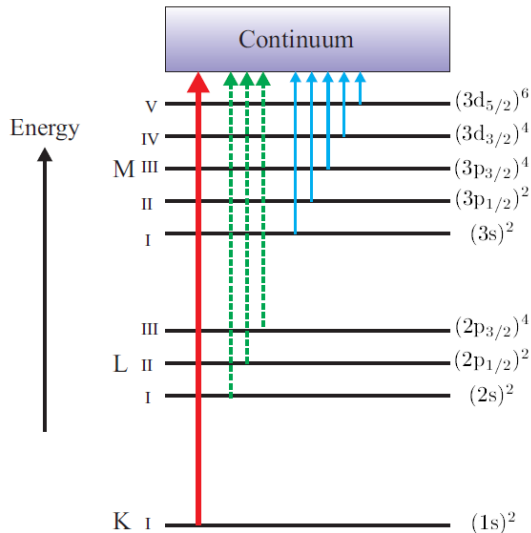


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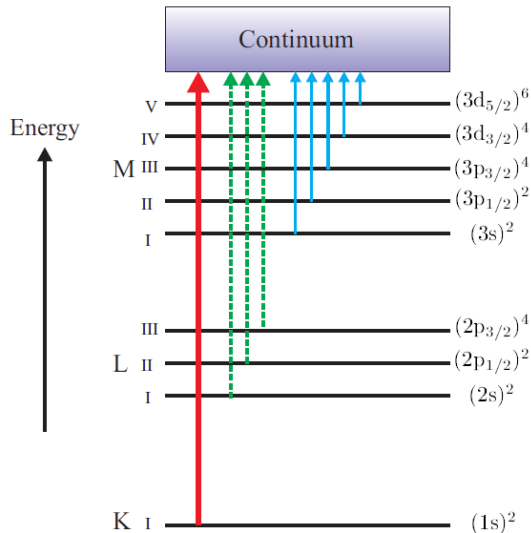
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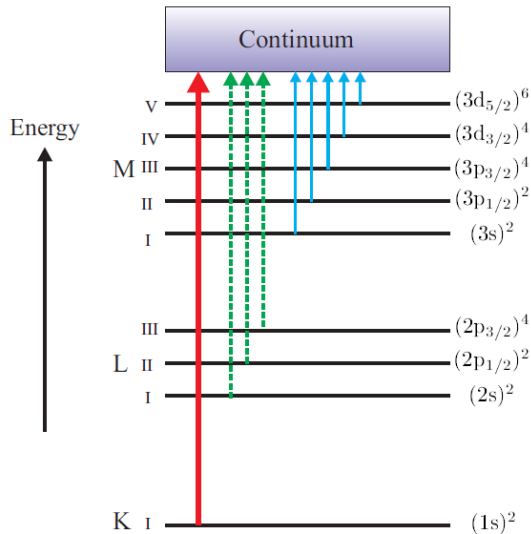
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Roman numerals increase from low to high values of l and j



Dirac bra-ket notation



Photoelectric absorption and inelastic scattering are quantum phenomena which cannot be treated semi-classically as scattering can.

Dirac bra-ket notation



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Paul Dirac developed a formalism for quantum mechanics which is commonly used. One part of this formalism is a compact notation which simplifies writing expectation value integrals. We will use this “bra-ket” notation when discussing photoabsorption.

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$$\psi^*(x)$$

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expectation value	$\int \psi^* Q \psi dx$	$\langle\psi Q \psi\rangle$	operator is applied to the right

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From first-order perturbation theory, the absorption cross section is given by

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$$\sigma_a = \frac{2\pi}{\hbar c} \frac{V^2}{4\pi^3} \int |M_{if}|^2 \delta(\mathcal{E}_f - \mathcal{E}_i) q^2 \sin \theta dq d\theta d\varphi$$

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$$\mathcal{H}_I = \frac{e\vec{p} \cdot \vec{A}}{m} + \frac{e^2 A^2}{2m}$$



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The first term gives **absorption**

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The first term gives **absorption** while the second produces **Thomson scattering** so we take only the first into consideration now.

$$\vec{A} = \hat{\varepsilon} \sqrt{\frac{\hbar}{2\epsilon_0 V \omega}} \left[a_k e^{i\vec{k} \cdot \vec{r}} + a_k^\dagger e^{-i\vec{k} \cdot \vec{r}} \right]$$

Free electron approximation



In order to evaluate the M_{if} matrix element we define the initial and final states

Free electron approximation



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$$M_{if} = \frac{e}{m} \sqrt{\frac{\hbar}{2\epsilon_0 V \omega}} \left[{}_e\langle 1|_\gamma \langle 0| (\vec{p} \cdot \hat{\epsilon}) a e^{i\vec{k} \cdot \vec{r}} + (\vec{p} \cdot \hat{\epsilon}) a^\dagger e^{-i\vec{k} \cdot \vec{r}} |1\rangle_\gamma |0\rangle_e \right]$$

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The calculation is simplified if the interaction Hamiltonian is applied to the left since the final state has only a free electron and no photon

Free electron approximation



The free electron state is an eigenfunction of the electron momentum operator

Free electron approximation



$${}_e\langle 1 | \vec{p} = (\hbar \vec{q}) {}_e\langle 1 |$$

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Photoelectron integral



$$M_{if} = \frac{e\hbar}{m} \sqrt{\frac{\hbar}{2\epsilon_0 V \omega}} (\vec{q} \cdot \hat{\epsilon}) \int \psi_f^* e^{i\vec{k} \cdot \vec{r}} \psi_i d\vec{r}$$



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which is the Fourier transform of the initial state 1s electron wave function

Photoelectron cross-section



the overall matrix element squared for a particular photoelectron final direction (φ, θ) is



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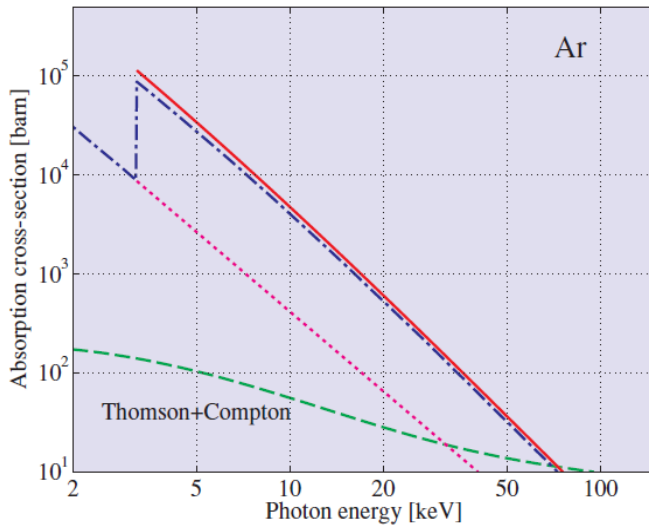
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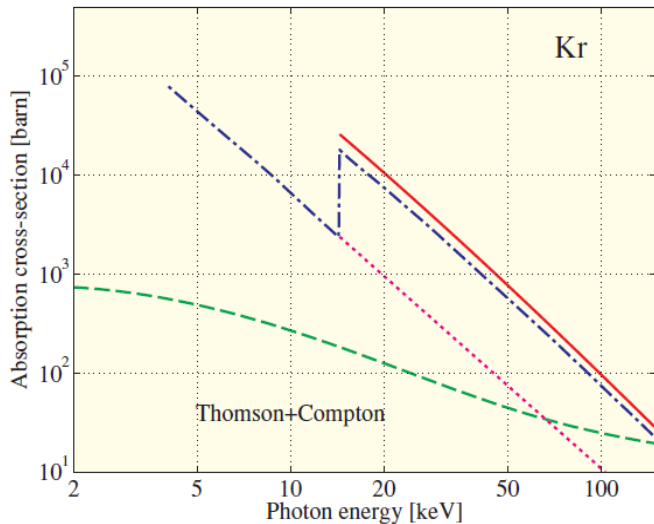
where the integral I_3 is given by

$$I_3 = \int \phi^2(\vec{Q}) q^2 \sin^2 \theta \cos^2 \varphi \delta(\mathcal{E}_f - \mathcal{E}_i) q^2 \sin \theta dq d\theta d\phi$$

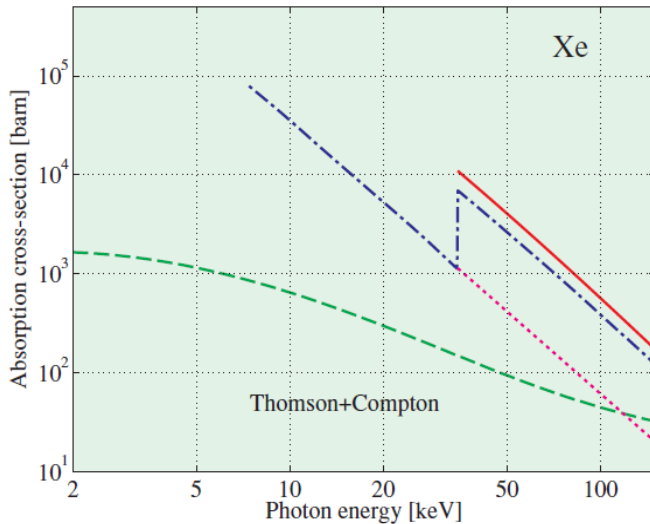
Calculated cross sections



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What is XAFS?



X-ray Absorption Fine-Structure (XAFS) is the modulation of the x-ray absorption coefficient at energies near and above an x-ray absorption edge. XAFS is also referred to as X-ray Absorption Spectroscopy (XAS) and is broken into 2 regimes:

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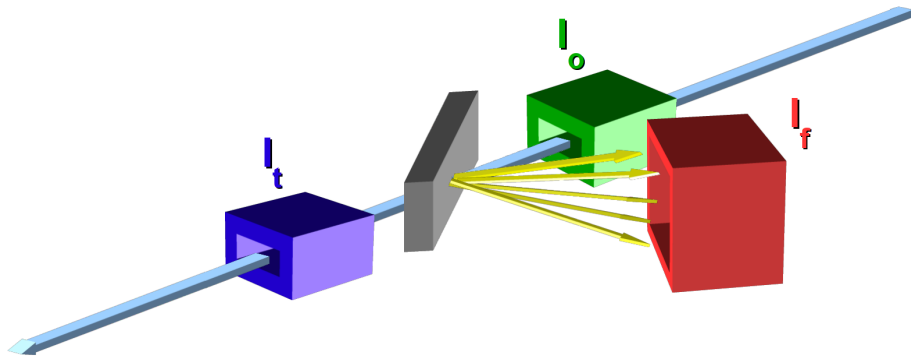
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- works at low concentrations
- has minimal sample requirements

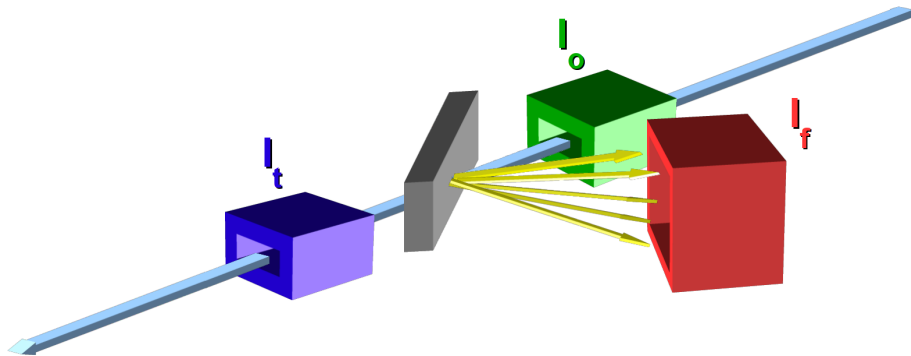
The EXAFS experiment



The EXAFS experiment



I_o = incident intensity

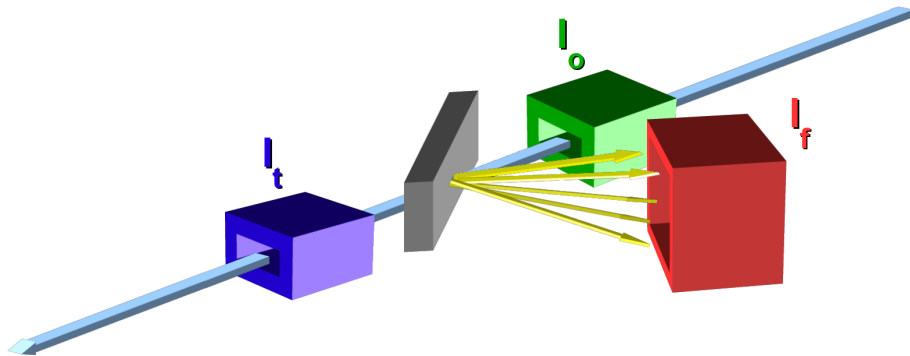


The EXAFS experiment



I_o = incident intensity

I_t = transmitted intensity



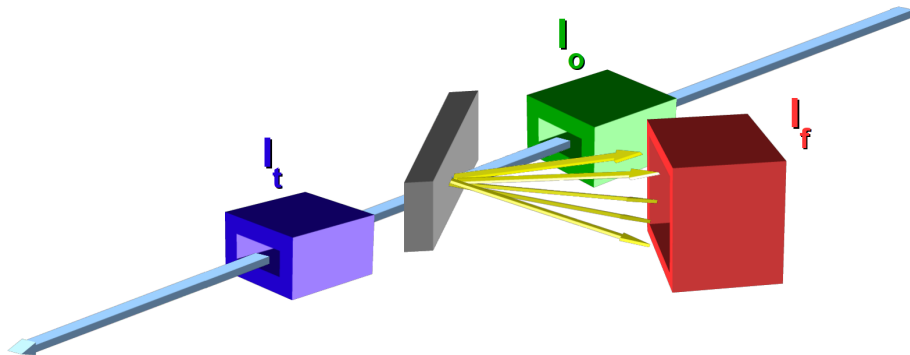
The EXAFS experiment



I_o = incident intensity

I_t = transmitted intensity

I_f = fluorescence intensity



The EXAFS experiment

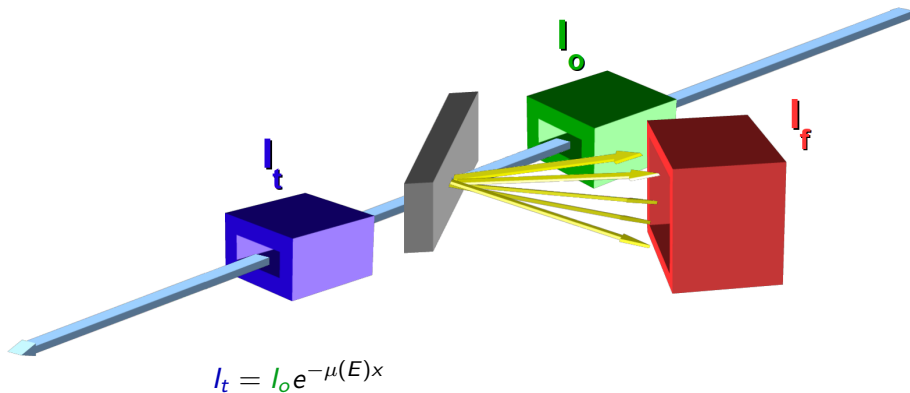


I_o = incident intensity

I_t = transmitted intensity

I_f = fluorescence intensity

x = sample thickness



The EXAFS experiment



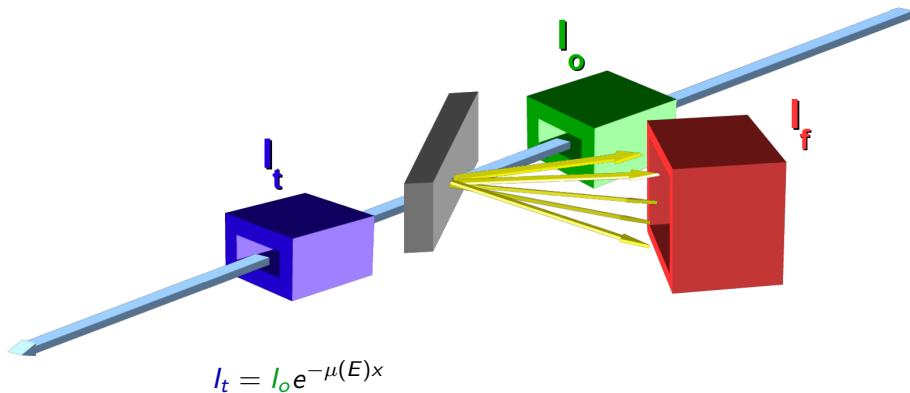
I_o = incident intensity

I_t = transmitted intensity

I_f = fluorescence intensity

x = sample thickness

$\mu(E)$ = absorption coefficient



The EXAFS experiment



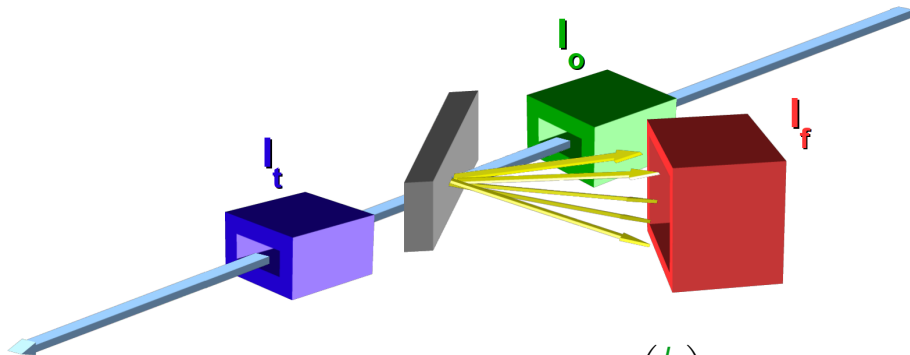
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$$I_t = I_o e^{-\mu(E)x} \longrightarrow \mu(E)x = \ln \left(\frac{I_o}{I_t} \right)$$

The EXAFS experiment



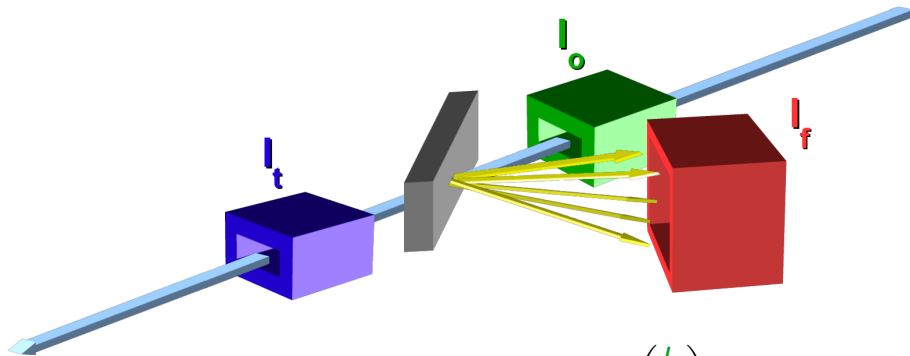
I_o = incident intensity

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I_f = fluorescence intensity

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$\mu(E)$ = absorption coefficient



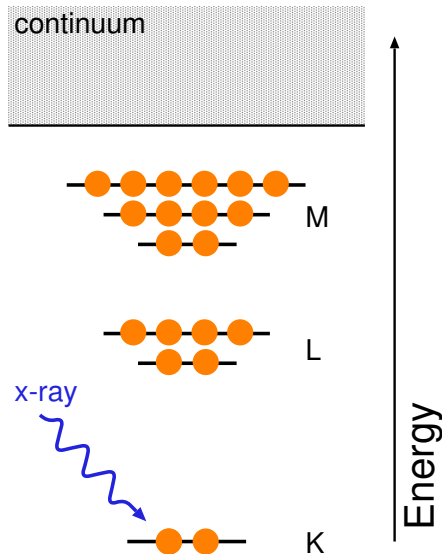
$$I_t = I_o e^{-\mu(E)x} \longrightarrow \mu(E)x = \ln\left(\frac{I_o}{I_t}\right)$$

$$\mu(E) \propto \frac{I_f}{I_o}$$

The x-ray absorption process



An **x-ray** is absorbed by an atom when the energy of the x-ray is transferred to a **core-level electron** (K , L , or M shell).

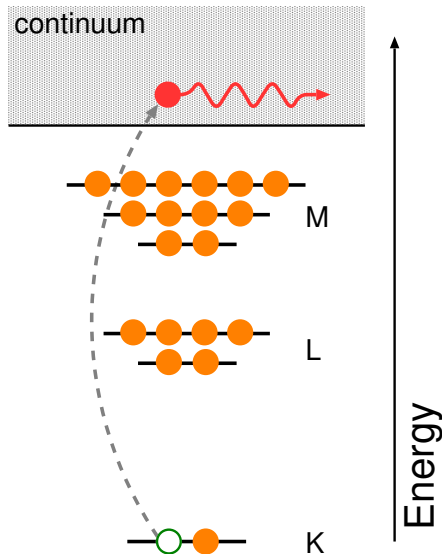


The x-ray absorption process



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The atom is in an excited state with an empty electronic level: a **core hole**.



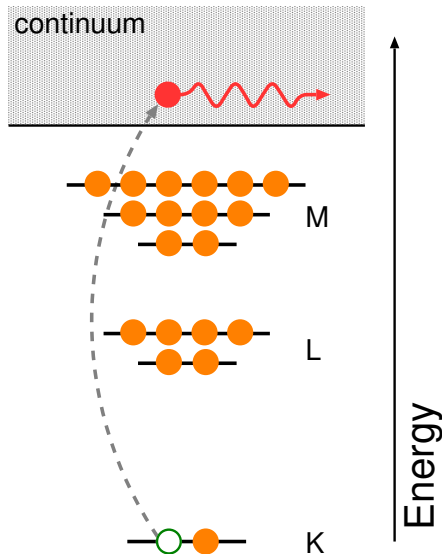
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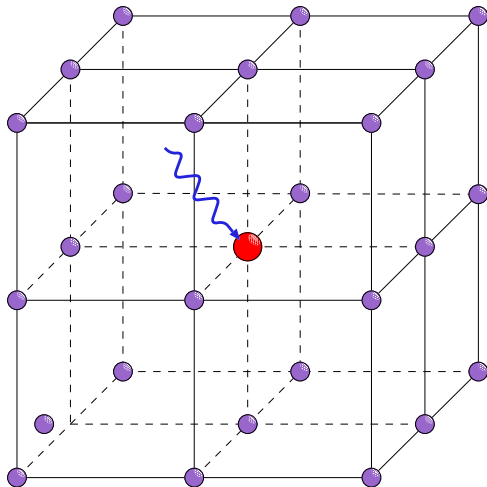
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Any excess energy from the x-ray is given to an ejected **photoelectron**, which expands as a spherical wave



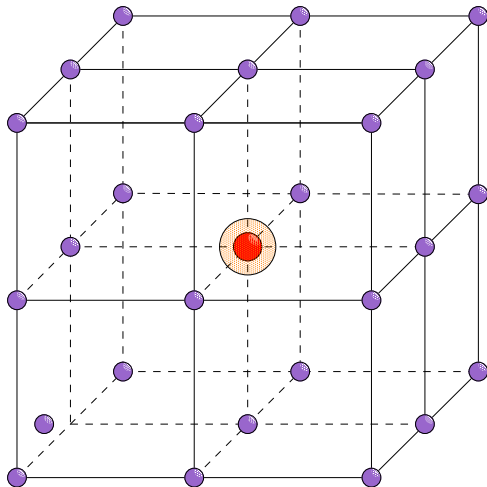
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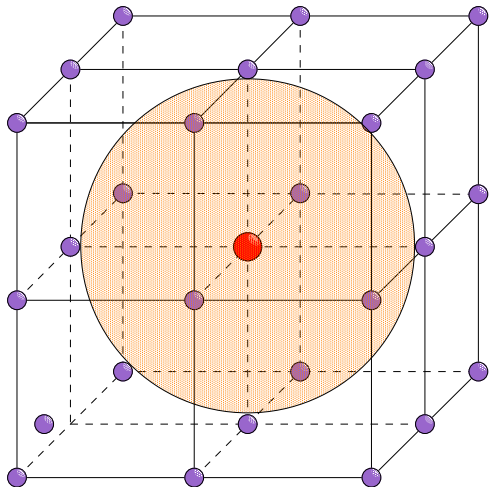
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Any excess energy from the x-ray is given to an ejected **photoelectron**, which expands as a spherical wave, reaches the neighboring electron clouds



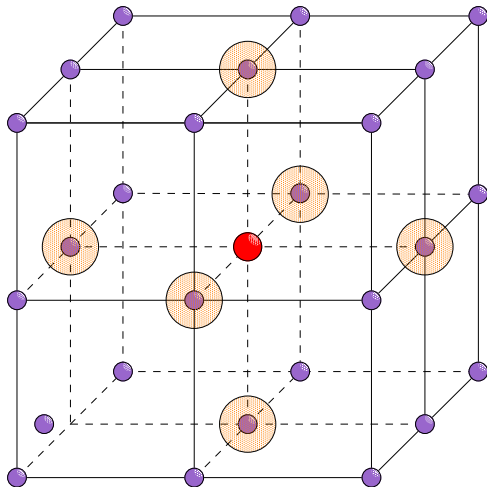
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Any excess energy from the x-ray is given to an ejected **photoelectron**, which expands as a spherical wave, reaches the neighboring electron clouds, and scatters back to the core hole



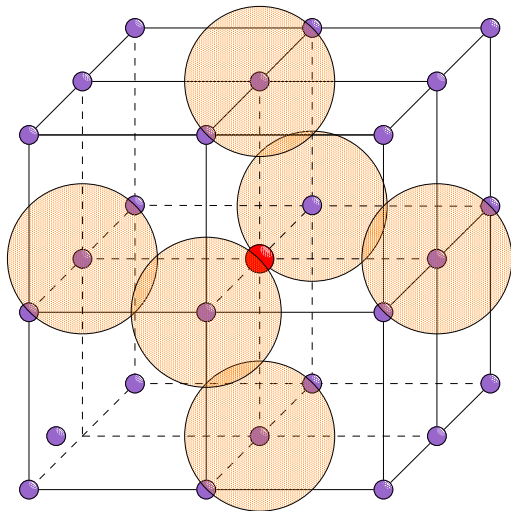
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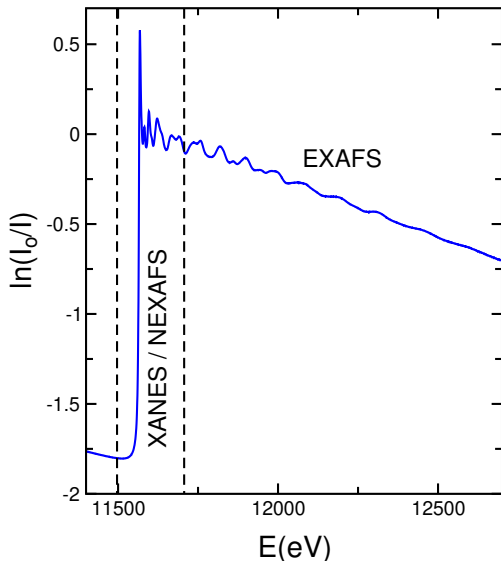
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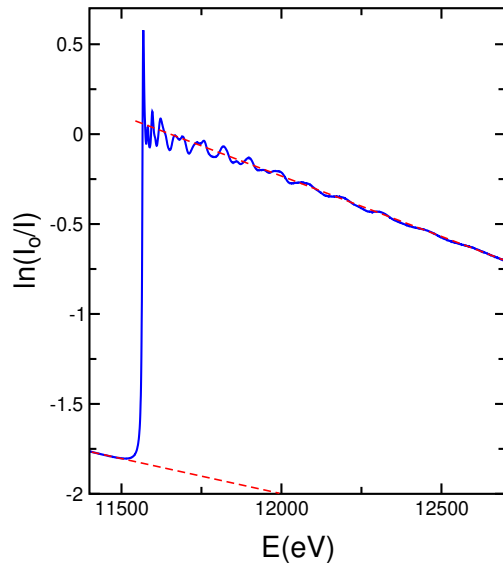
Any excess energy from the x-ray is given to an ejected **photoelectron**, which expands as a spherical wave, reaches the neighboring electron clouds, and scatters back to the core hole, creating interference patterns called XANES and EXAFS.



EXAFS data extraction



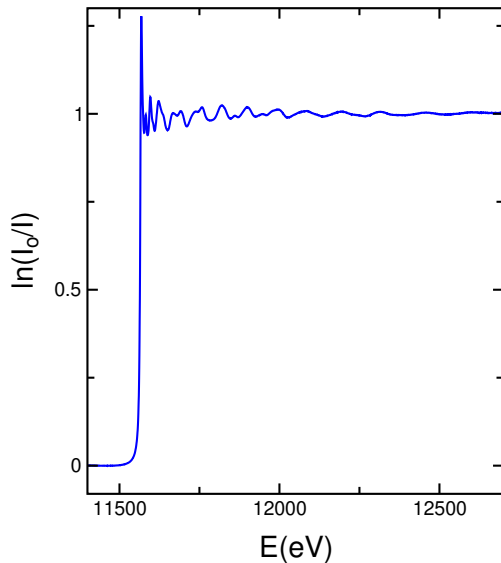
normalize by fitting **pre-edge** and **post-edge** trends



EXAFS data extraction



normalize by fitting pre-edge and post-edge trends

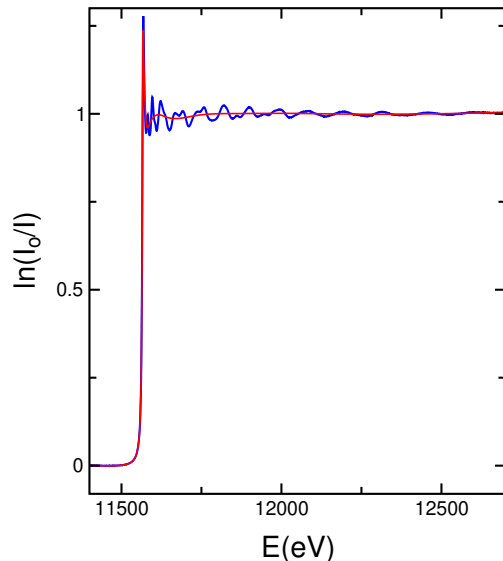


EXAFS data extraction



normalize by fitting pre-edge and post-edge trends

remove “smooth” μ_0 background



EXAFS data extraction

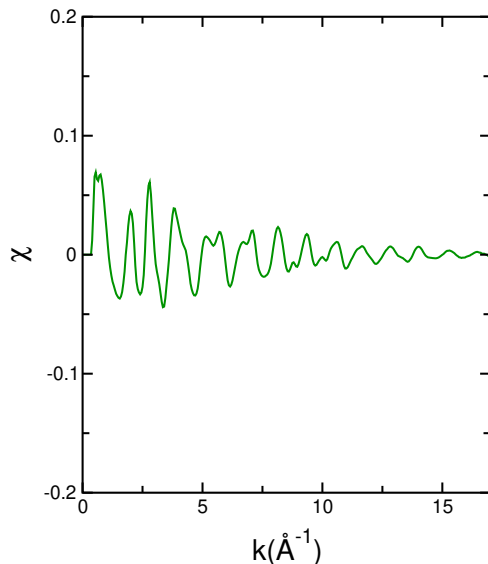


normalize by fitting **pre-edge** and **post-edge** trends

remove “smooth” μ_0 background

convert to **photoelectron momentum** space

$$k = \frac{2\pi}{hc} \sqrt{\mathcal{E} - \mathcal{E}_0}$$



EXAFS data extraction



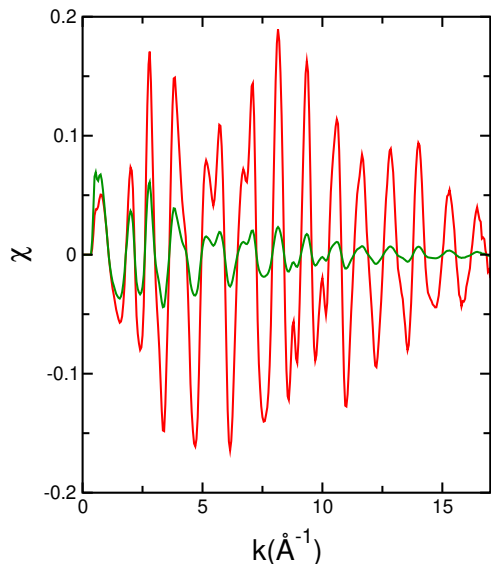
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weight by appropriate power of k to obtain “good” envelope which clearly shows that EXAFS is a sum of oscillations with varying frequencies and phases



EXAFS data extraction



normalize by fitting **pre-edge** and **post-edge** trends

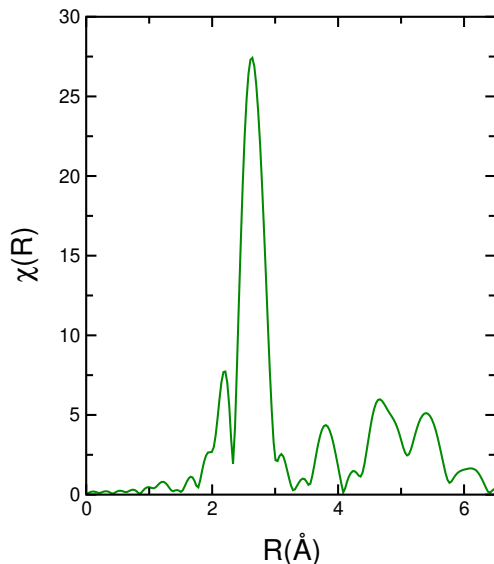
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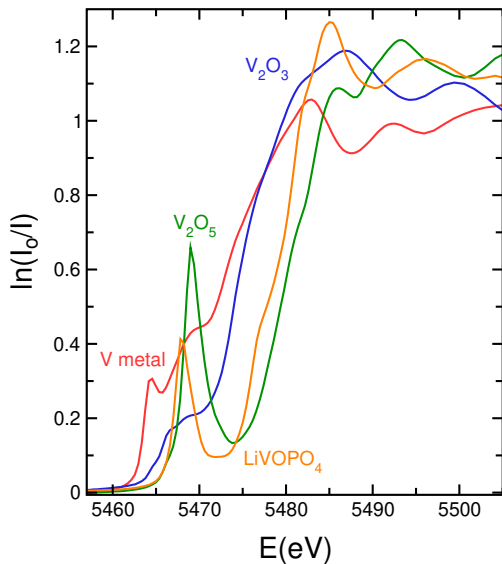
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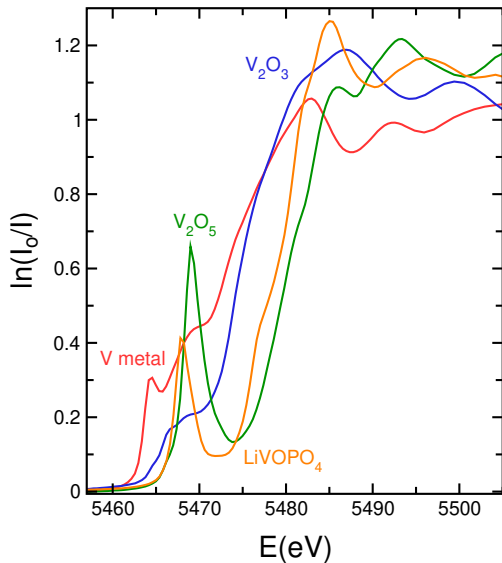
Fourier transform to get **real space EXAFS**



XANES edge shifts and pre-edge peaks

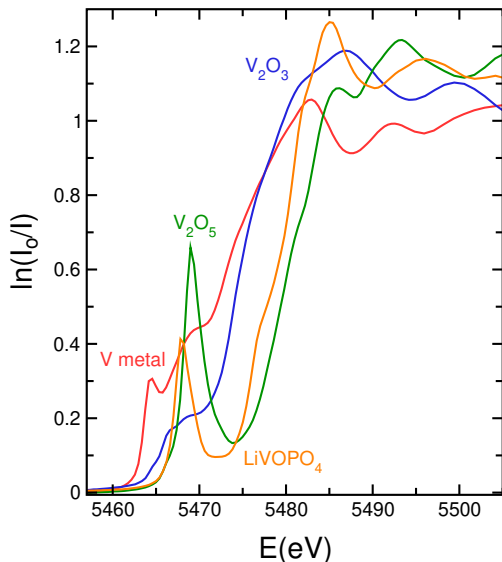


XANES edge shifts and pre-edge peaks



The shift of the edge position can be used to determine the valence state.

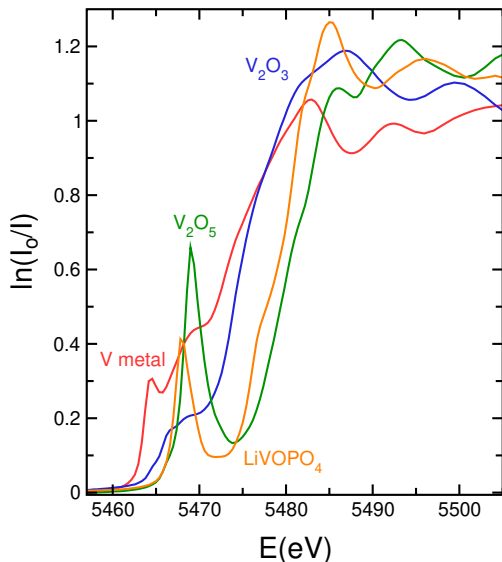
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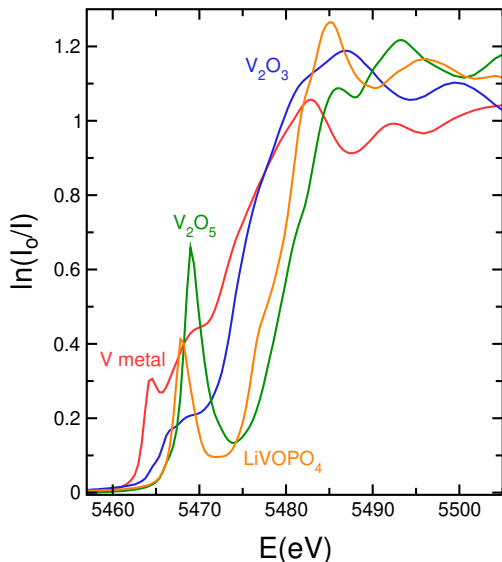


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XANES can be used as a fingerprint of phases and XANES analysis can be as simple as making linear combinations of “known” spectra to get composition.

Modern codes, such as FEFF9, are able to accurately compute XANES features.