



• Information about:



- Information about:
  - (a) Final presentation



- Information about:
  - (a) Final presentation
  - (b) Final project



- Information about:
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- SAXS papers



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- Lattice & basis functions



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Reading Assignment: Chapter 5.2–5.3



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Homework Assignment #04:

Chapter 4: 2,4,6,7.10

due Tuesday, October 19, 2021



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Reading Assignment: Chapter 5.2–5.3

Homework Assignment #04:

Chapter 4: 2,4,6,7.10

due Tuesday, October 19, 2021

Homework Assignment #05:

Chapter 5: 1,3,7,9,10

due Tuesday, November 02, 2021



1. Choose paper for presentation



2/29

- 1. Choose paper for presentation
- 2. Clear it with me!



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- 3. Do some background research on the technique



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- 4. Prepare a 15 minute presentation



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- 4. Prepare a 15 minute presentation
- 5. Be ready for questions!



3/29

1. Come up with a potential experiment



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- Make sure to give reasonable answers forall the questions



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- Find appropriate beamline(s) and if needed contact the beamline scientists (they are used to it)
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- Make sure to give reasonable answers forall the questions
- 7. Put me as one of the investigators of the proposal



The SAXS scattered intensity from a dilute solution depends on the single particle form factor,  $\mathcal{F}(\vec{Q})$ , the volume of the particle,  $V_p$ , and the density difference from the solvent,  $\Delta a = (a_1, \dots, a_{r-1})$ 

$$\Delta 
ho = (
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 $I^{SAXS}(Q) \approx \Delta \rho^2 V_p^2 e^{-Q^2 R_g^2/3}$ 

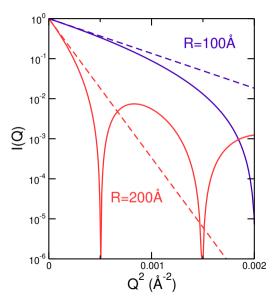
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4/29

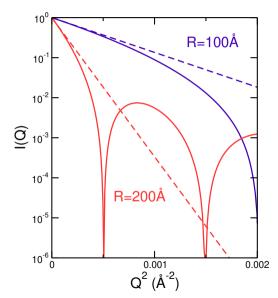


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$$R_g^2 = \frac{\int_{V_p} \rho_{sl,p}(\vec{r}) r^2 dV_p}{\int_{V_p} \rho_{sl,p}(\vec{r}) dV_p}$$





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- SAXS of irradiated Zn nanoparticles
- Nucleation and growth of & glycine crystals

# SAXS of irradiated Zn nanoparticles



6/29

Zn nanoparticles formed in  $SiO_2$  by ion implantation irradiated with high energy  $Xe^{+14}$  ions.

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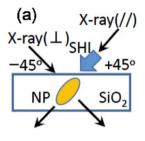
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Expt. geometry

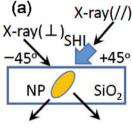
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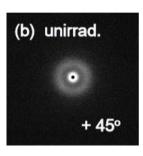
6/29

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Expt. geometry



Unirradiated

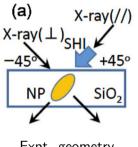
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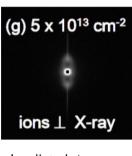
6/29

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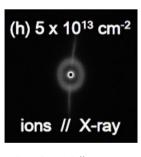
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Irradiated  $\perp x$ -rays



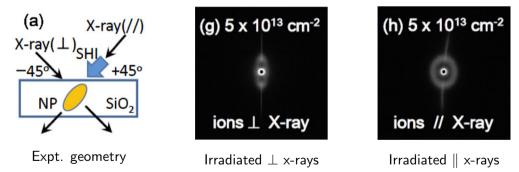
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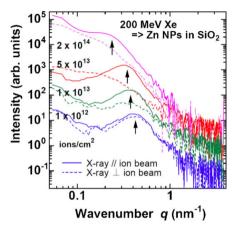
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Straight lines from ion tracks, seen in both directions and which persist to the highest fluences.

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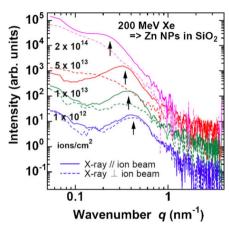


Interference peak persists for  $\parallel$  but not  $\perp$  incidence

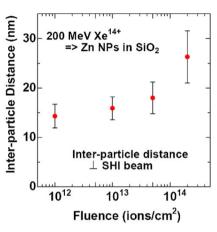
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7/29



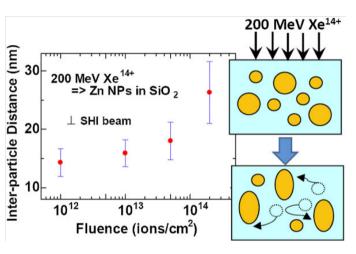
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Interparticle distance increases as a function of irradiation fluence

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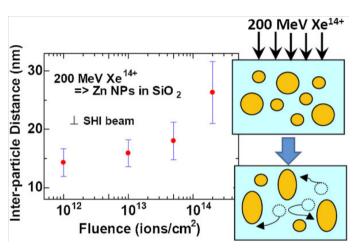




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Growth of interparticle spacing is due to dissolution and re-agglomeration with fluence leading to larger interparticle spacings



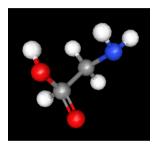
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Can SAXS help us understand the nucleation and growth of a simple molecule which is the prototype for pharmaceutical compounds?

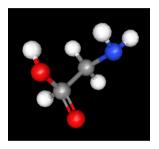


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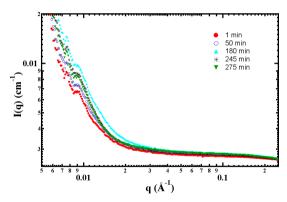




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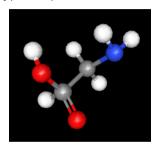


Initial studies at 12keV observe change in  $R_g$  upon crystallization.

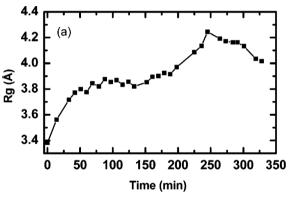




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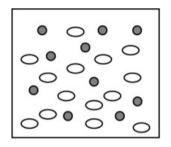
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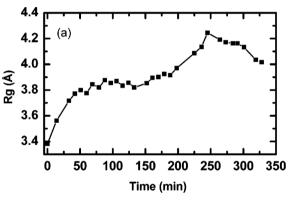
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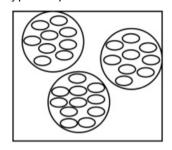
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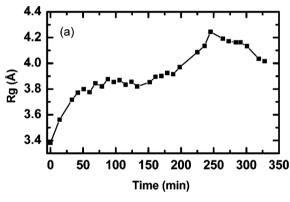
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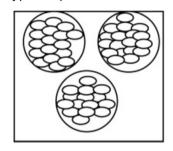
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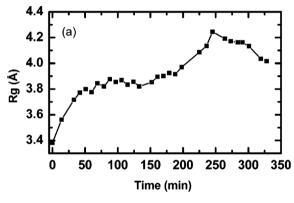
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Change to 25 keV x-rays reduces crystallization time to under 90 min



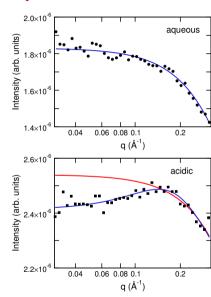
10 / 29

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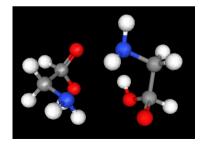




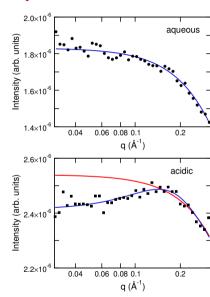
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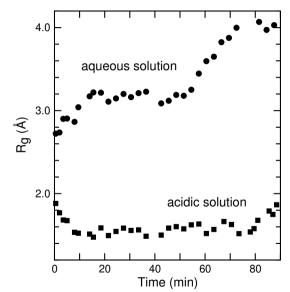
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# Glycine $R_g$



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In aqueous solution,  $R_g$  implies dimerization and increases due to aggregation until crystallization

Carlo Segre (Illinois Tech) PHYS 570 - Fall 2021 October 05, 2021

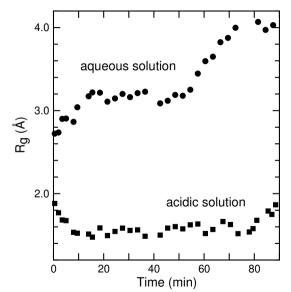
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D. Erdemir et al. Phys. Rev. Lett. 99, 115702 (2007)

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11/29



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In acidic solution, Rg remains small and implies that no dimerization or aggregation occurs before nucleation

Carlo Segre (Illinois Tech) PHYS 570 - Fall 2021 October 05, 2021

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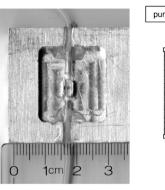
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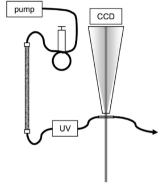
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Mathew, Mirza & Menhart, "Liquid-chromatography-coupled SAXS for accurate sizing of aggregating proteins," *J. Synchrotron Rad.* **11**, 314-318 (2004) developed a technique which is now being used routinely in biological SAXS, called Size Exclusion Chromatography SAXS.



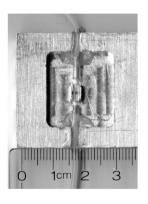


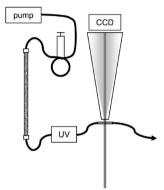


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2m SAXS camera,  $1.03\text{\normalfont\AA}\ (12\ \text{keV})\ \text{x-}$  rays were used





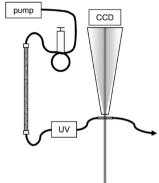
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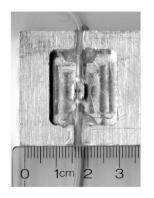
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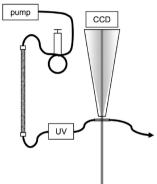


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samples of (1) cytochrome c, (2) plasminogen, (3) mixture of cytochrome c bovine serum albumin, and blue dextran





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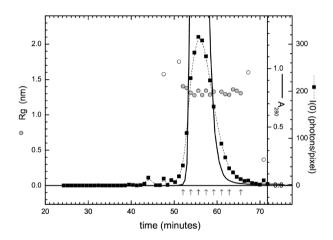


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for cytochrome c, forward scatter (black squares) measures the total number of electrons in the beam



<sup>&</sup>quot;Liquid-chromatography-coupled SAXS for accurate sizing of aggregating proteins," Mathew, Mirza & Menhart, J. Synchrotron Rad. 11, 314-318 (2004).

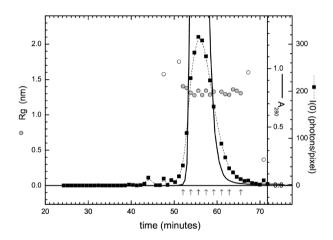


2m SAXS camera, 1.03Å  $(12 \text{ keV}) \times \text{rays}$  were used

2s exposure times every 20s, with 0.25 ml/min flow rate

samples of (1) cytochrome c, (2) plasminogen, (3) mixture of cytochrome c bovine serum albumin, and blue dextran

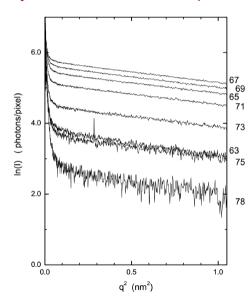
for cytochrome c, forward scatter (black squares) measures the total number of electrons in the beam  $R_g$  is constant throughout the main peak



<sup>&</sup>quot;Liquid-chromatography-coupled SAXS for accurate sizing of aggregating proteins," Mathew, Mirza & Menhart, J. Synchrotron Rad. 11, 314-318 (2004).

#### Cytochrome c - Guinier plots



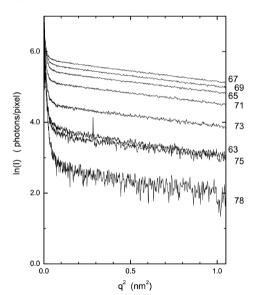


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Carlo Segre (Illinois Tech)

### Cytochrome c - Guinier plots





Plot from times marked with arrows on  $R_{\rm g}$  plot.

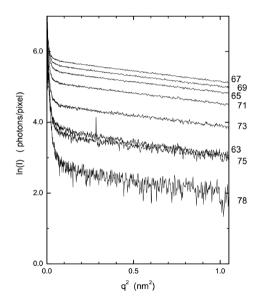
"Liquid-chromatography-coupled SAXS for accurate sizing of aggregating proteins," Mathew, Mirza & Menhart, *J. Synchrotron Rad.* 11, 314-318 (2004).

Carlo Segre (Illinois Tech)

PHYS 570 - Fall 2021

### Cytochrome c - Guinier plots





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Guinier plots are parallel, indicating a single species present (a single critical exponent).

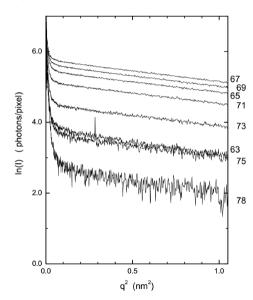
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PHYS 570 - Fall 2021

# Cytochrome c - Guinier plots





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Guinier plots are parallel, indicating a single species present (a single critical exponent).

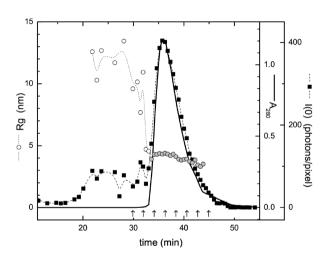
Even lowest intensity data set gives a consistent  $R_g$ .

"Liquid-chromatography-coupled SAXS for accurate sizing of aggregating proteins," Mathew, Mirza & Menhart, *J. Synchrotron Rad.* 11, 314-318 (2004).

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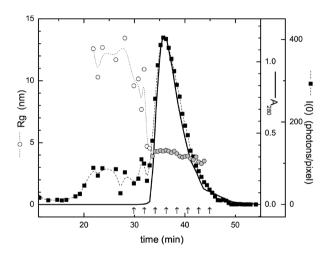


Constant  $R_g$  in region where  $A_{UV}/I(0)$  is constant.





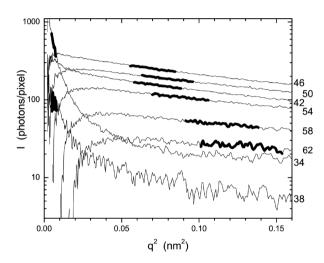
Constant  $R_g$  in region where  $A_{UV}/I(0)$  is constant. Aggregates preced the main peak and show wildly varying  $R_g$ .





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Guinier plots labeled 34 and 38 show presence of aggregates and the slopes are not parallel, indicating multiple sized species

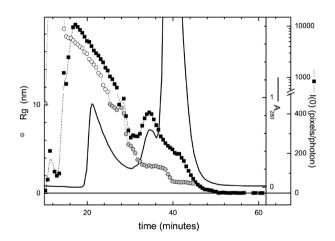




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The three components show consistent  $R_g$  and can be individually identified despite the overlap.



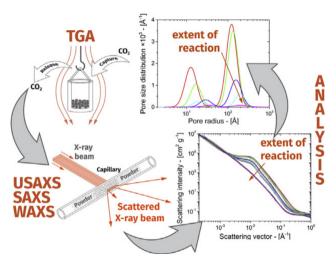


SAXS was used to study the nature of the porosity and particle sizes of CaO obtained by calcining CaCO<sub>3</sub>.

<sup>&</sup>quot;Analysis of textural properties of CaO-based CO<sub>2</sub> sorbents by ex-situ USAXS," A. Benedetti, J. Ilavsky, C.U. Segre, and M. Strumendo, *Chem. Eng. J.* 355, 760-776 (2019).



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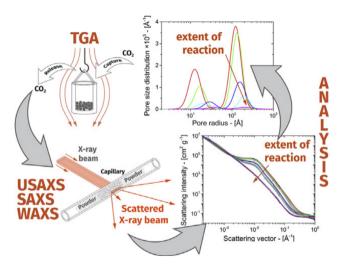


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CaO can be used for carbon capture and then recycled by calcination. It is important to understand the meso structure of the material at different stages of the process



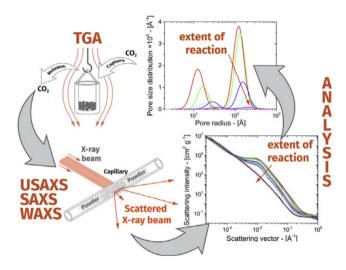
<sup>&</sup>quot;Analysis of textural properties of CaO-based CO<sub>2</sub> sorbents by ex-situ USAXS," A. Benedetti, J. Ilavsky, C.U. Segre, and M. Strumendo, *Chem. Eng. J.* 355, 760-776 (2019).



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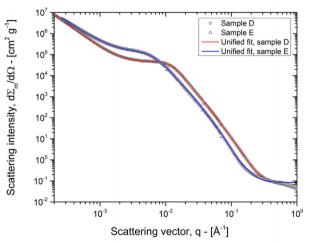
The samples were studied ex-situ at Sector 9-ID using USAXS and analyzed with a unified fit model



<sup>&</sup>quot;Analysis of textural properties of CaO-based CO<sub>2</sub> sorbents by ex-situ USAXS," A. Benedetti, J. Ilavsky, C.U. Segre, and M. Strumendo, *Chem. Eng. J.* 355, 760-776 (2019).



Sample D was calcined at  $900 \, ^{\circ}\text{C}$  for 50 minutes while sample E was calcined at the same temperature for  $240 \, \text{minutes}$ 

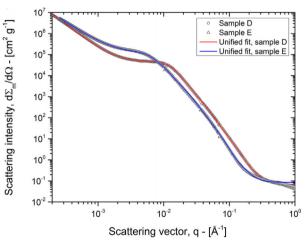


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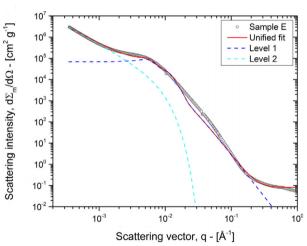
The SAXS shows the grain growth evolution between the two samples and it is clear that the samples need a multilevel unified fit



<sup>&</sup>quot;Analysis of textural properties of CaO-based CO<sub>2</sub> sorbents by ex-situ USAXS," A. Benedetti, J. Ilavsky, C.U. Segre, and M. Strumendo, *Chem. Eng. J.* 355, 760-776 (2019).



The components of the unified fit model are shown for a two level fit and it is clear that 2 levels are insufficient.

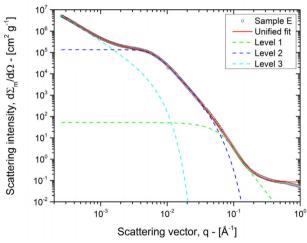


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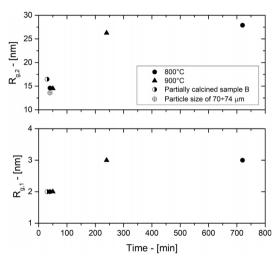
A three level fit works well for the calcined samples and from this one can extract the pore sizes for two different pore populations in the calcined samples



<sup>&</sup>quot;Analysis of textural properties of CaO-based CO<sub>2</sub> sorbents by ex-situ USAXS," A. Benedetti, J. Ilavsky, C.U. Segre, and M. Strumendo, *Chem. Eng. J.* 355, 760-776 (2019).



Fitting a series of samples calcined at varying temperatures and times shows the evolution of the radii of gyration of the two populations corresponding to the pore sizes

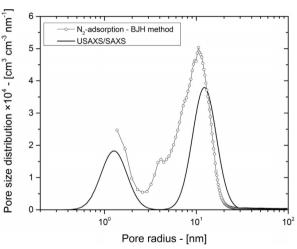


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The resulting pore size distributions correspond well to those measured using gas adsorption methods



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In this case, the distances probed are similar to those in liquid scattering but the sample has an ordered lattice which results in very prominent diffraction peaks separated by ranges with zero scattered intensity.

We will now proceed to develop a model for this kind of scattering starting with some definitions in 2D space.



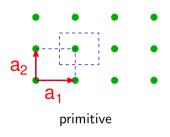
$$\vec{R}_n = n_1 \vec{a}_1 + n_2 \vec{a}_2$$

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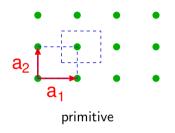


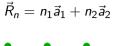
21 / 29

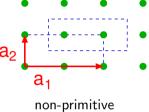
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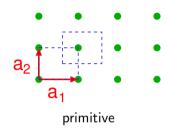




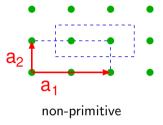


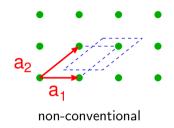




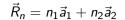


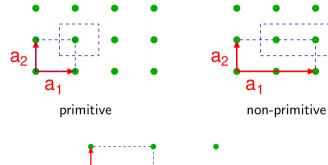
$$\vec{R}_n = n_1 \vec{a}_1 + n_2 \vec{a}_2$$

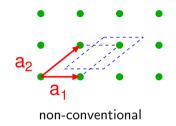










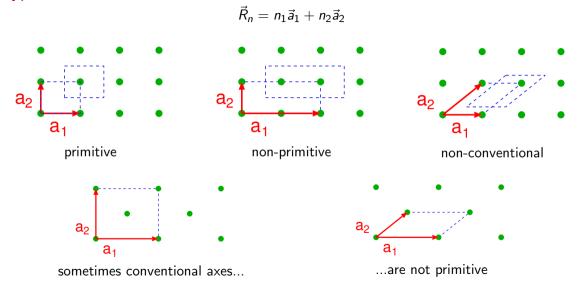




sometimes conventional axes...

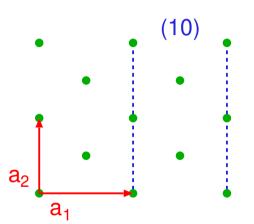


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#### Miller indices



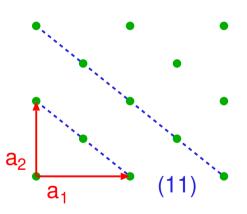


planes designated (hk), intercept the unit cell axes at

$$\frac{a_1}{h}, \frac{a_2}{k}$$

#### Miller indices



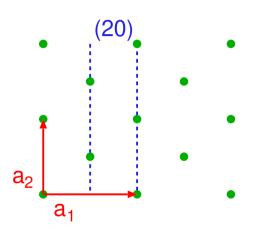


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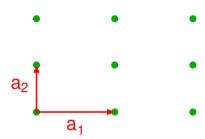
$$\frac{a_1}{h}, \frac{a_2}{k}$$

for a lattice with orthogonal unit vectors

$$\frac{1}{d_{hk}^2} = \frac{h^2}{a_1^2} + \frac{k^2}{a_2^2}$$

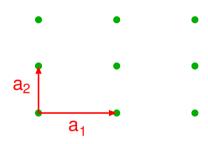
# Reciprocal lattice





## Reciprocal lattice





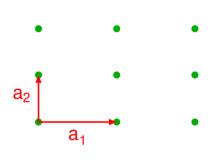
$$\vec{a}_1^* = \frac{2\pi}{V_c} \vec{a}_2 \times \vec{a}_3 \qquad \vec{a}_2^* = \frac{2\pi}{V_c} \vec{a}_3 \times \vec{a}_1 \qquad \vec{a}_3^* = \frac{2\pi}{V_c} \vec{a}_1 \times \vec{a}_2$$

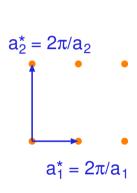
$$\vec{a}_2^* = \frac{2\pi}{V_-} \vec{a}_3 \times \vec{a}_1$$

$$\vec{a}_3^* = \frac{2\pi}{V} \vec{a}_1 \times \vec{a}_2$$

## Reciprocal lattice







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$$ec{a}_3^* = rac{2\pi}{V} ec{a}_1 imes ec{a}_2$$



If the basis of a one-dimensional system is described by the function  $\mathcal{B}(x)$  then the crystal is described by the function



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$$\mathcal{C}(x) = \sum_n \mathcal{B}(x - na)$$



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$$C(x) = \sum_{n} \mathcal{B}(x - na)$$

the lattice, which is a collection of points in space, can be written



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$$\mathcal{L}(x) = \sum_{n} \delta(x - na)$$



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convoluting the lattice and basis function we write



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$$\mathcal{L}(x) \star \mathcal{B}(x) = \int_{-\infty}^{\infty} \mathcal{L}(x') \mathcal{B}(x - x') dx'$$



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$$= \sum_{n} \int_{-\infty}^{\infty} \delta(x' - na) \mathcal{B}(x - x') dx' = \sum_{n} \mathcal{B}(x - na) = \mathcal{C}(x)$$



$$F^{crystal}(\vec{Q}) = \sum_{l}^{N} f_{l}(\vec{Q}) e^{i\vec{Q}\cdot\vec{r}_{l}}$$



$$F^{crystal}(\vec{Q}) = \sum_{l}^{N} f_l(\vec{Q}) e^{i\vec{Q}\cdot\vec{r_l}} = \sum_{\vec{R}_n + \vec{r_i}}^{N} f_j(\vec{Q}) e^{i\vec{Q}\cdot(\vec{R}_n + \vec{r_j})}$$



$$egin{aligned} F^{crystal}(ec{Q}) &= \sum_{l}^{N} f_{l}(ec{Q}) e^{i ec{Q} \cdot ec{r}_{l}} = \sum_{ec{R}_{n} + ec{r}_{j}}^{N} f_{j}(ec{Q}) e^{i ec{Q} \cdot (ec{R}_{n} + ec{r}_{j})} \ &= \sum_{i} f_{j}(ec{Q}) e^{i ec{Q} \cdot ec{r}_{j}} \sum_{n} e^{i ec{Q} \cdot ec{R}_{n}} \end{aligned}$$



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$$= \sum_{i} f_{j}(\vec{Q}) e^{i\vec{Q}\cdot\vec{r_{j}}} \sum_{n} e^{i\vec{Q}\cdot\vec{R}_{n}} = F^{unit\ cell} F^{lattice}$$



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Since  $F^{crystal}(\vec{Q})$  is simply the Fourier Transform of the crystal function,  $C(x) = \mathcal{L}(x) \star \mathcal{B}(x)$ , it must be the product of the Fourier Transforms of  $\mathcal{L}(x)$  and  $\mathcal{B}(x)$ .



$$F^{crystal}(\vec{Q}) = \sum_{l}^{N} f_{l}(\vec{Q}) e^{i\vec{Q}\cdot\vec{r}_{l}} = \sum_{\vec{R}_{n}+\vec{r}_{j}}^{N} f_{j}(\vec{Q}) e^{i\vec{Q}\cdot(\vec{R}_{n}+\vec{r}_{j})}$$

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$$\vec{Q} \cdot \vec{R}_n = 2\pi m, \quad m = \text{integer}$$



$$F^{crystal}(\vec{Q}) = \sum_{l}^{N} f_{l}(\vec{Q}) e^{i\vec{Q}\cdot\vec{r_{l}}} = \sum_{\vec{R}_{n}+\vec{r_{j}}}^{N} f_{j}(\vec{Q}) e^{i\vec{Q}\cdot(\vec{R}_{n}+\vec{r_{j}})}$$

$$= \sum_{j} f_{j}(\vec{Q}) e^{i\vec{Q}\cdot\vec{r_{j}}} \sum_{n} e^{i\vec{Q}\cdot\vec{R}_{n}} = F^{unit \ cell} F^{lattice}$$

$$\vec{Q} \cdot \vec{R}_n = 2\pi m, \quad m = \text{integer}$$

$$\vec{G}_{hkl} = h\vec{a}_1^* + k\vec{a}_2^* + l\vec{a}_3^*, \quad h, k, l = \text{integer}$$



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Since  $F^{crystal}(\vec{Q})$  is simply the Fourier Transform of the crystal function,  $\mathcal{C}(x) = \mathcal{L}(x) \star \mathcal{B}(x)$ , it must be the product of the Fourier Transforms of  $\mathcal{L}(x)$  and  $\mathcal{B}(x)$ .  $F^{lattice}$  is a very large sum  $(\sim 10^{12})$  so the only time it gives values appreciably greater than 1 is when:

$$\vec{Q} \cdot \vec{R}_n = 2\pi m, \quad m = \text{integer}$$
  $\vec{G}_{hkl} = h\vec{a}_1^* + k\vec{a}_2^* + l\vec{a}_3^*, \quad h, k, l = \text{integer}$   $\vec{G}_{hkl} \cdot \vec{R}_n = (n_1\vec{a}_1 + n_2\vec{a}_2 + n_3\vec{a}_3) \cdot (h\vec{a}_1^* + k\vec{a}_2^* + l\vec{a}_3^*) = 2\pi (hn_1 + kn_2 + ln_3) = 2\pi m$ 

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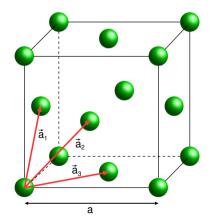


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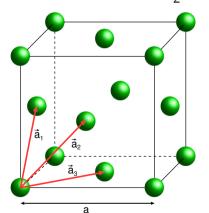
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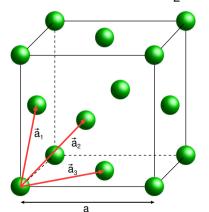


$$\vec{a}_1 = \frac{a}{2}(\hat{y} + \hat{z}),$$





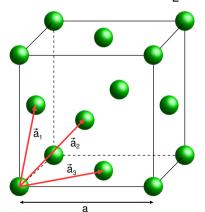
$$\vec{a}_1 = \frac{a}{2}(\hat{y} + \hat{z}), \quad \vec{a}_2 = \frac{a}{2}(\hat{z} + \hat{x}),$$





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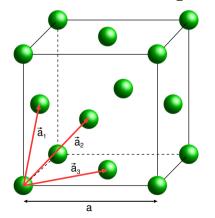
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The primitive lattice vectors of the face-centered cubic lattice are

$$\vec{a}_1 = \frac{a}{2}(\hat{y} + \hat{z}), \quad \vec{a}_2 = \frac{a}{2}(\hat{z} + \hat{x}), \quad \vec{a}_3 = \frac{a}{2}(\hat{x} + \hat{y})$$

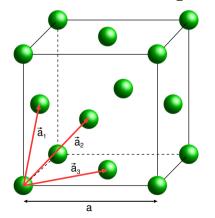




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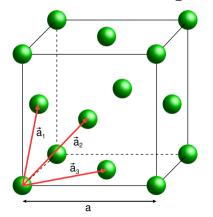
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$$ec{a}_1 = rac{a}{2}(\hat{y} + \hat{z}), \quad ec{a}_2 = rac{a}{2}(\hat{z} + \hat{x}), \quad ec{a}_3 = rac{a}{2}(\hat{x} + \hat{y})$$

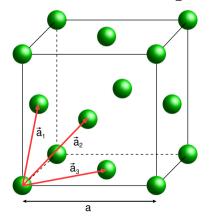


$$v_c = ec{a}_1 \cdot ec{a}_2 imes ec{a}_3 = ec{a}_1 \cdot rac{a^2}{4} \left( \hat{y} + \hat{z} - \hat{x} 
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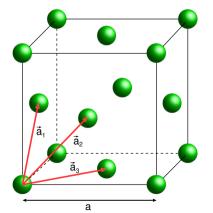


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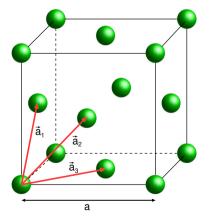


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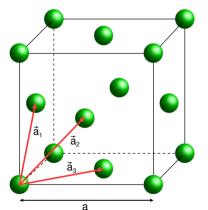


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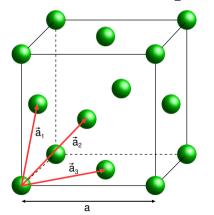


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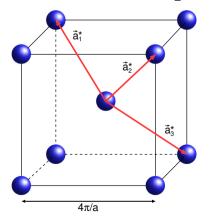
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The volume of the unit cell is

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which is a body-centered cubic lattice



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$$S_N(\vec{Q}) = \sum_n e^{i\vec{Q}\cdot\vec{R}_n} = \sum_{n=0}^{N-1} e^{iQna} \longrightarrow |S_N(Q)| = \frac{\sin(NQa/2)}{\sin(Qa/2)} = \frac{\sin(N[h+\xi]\pi)}{\sin([h+\xi]\pi)}$$
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if the Laue condition is not exactly fulfilled then  $Q = [h + \xi]a^*$  and the sum becomes the numerator can be simplified as

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since  $\delta(a^*\xi) = \delta(\xi)/a^*$ 



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the 1D modulus squared



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$$|S_N(Q)|^2 
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$$|S_{N}(Q)|^{2}
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a\*  $\sum_{G_{h}}\delta(Q-G_{h})$ 

in 2D, with  $N_1 \times N_2 = N$  unit cells



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$$\left|S_{\mathcal{N}}(\vec{Q})\right|^2 
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$$egin{align} \left|S_{\mathcal{N}}(\vec{Q})
ight|^2 &
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and similarly in 3D



the 1D modulus squared

$$|S_N(Q)|^2 o Na^* \sum_{G_h} \delta(Q - G_h)$$

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$$\left|S_{N}(\vec{Q})\right|^{2}
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